

Correlation between morphology and transport properties of quasi-free-standing monolayer graphene (QFMLG)

Yuya Murata¹, Torge Mashoff², Makoto Takamura³, Shinichi Tanabe³,
Hiroki Hibino³, Fabio Beltram^{1,2}, and Stefan Heun¹

1 NEST, Istituto Nanoscienze-CNR and Scuola Normale Superiore, Pisa, Italy

2 Center for Nanotechnology Innovation @ NEST, Istituto Italiano di Tecnologia, Pisa, Italy

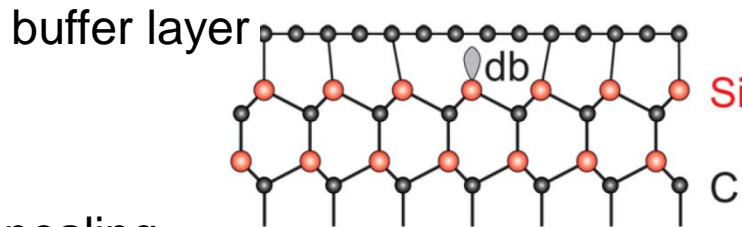
3 NTT Basic Research Laboratories, NTT Corporation, Japan

National Enterprise for nanoScience and nanoTechnology



Introduction

Graphene on silicon carbide (SiC) (0001)

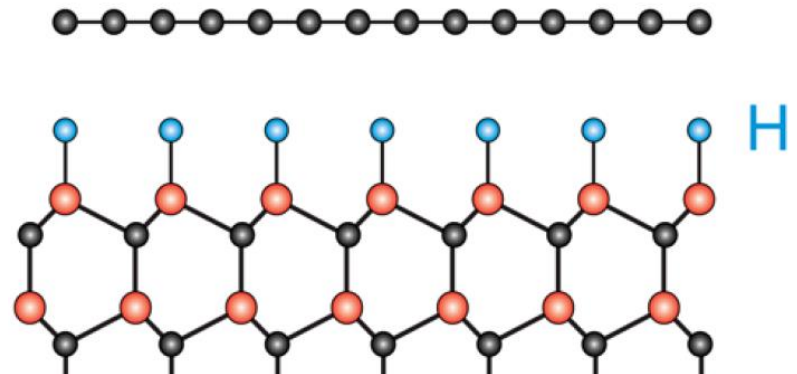
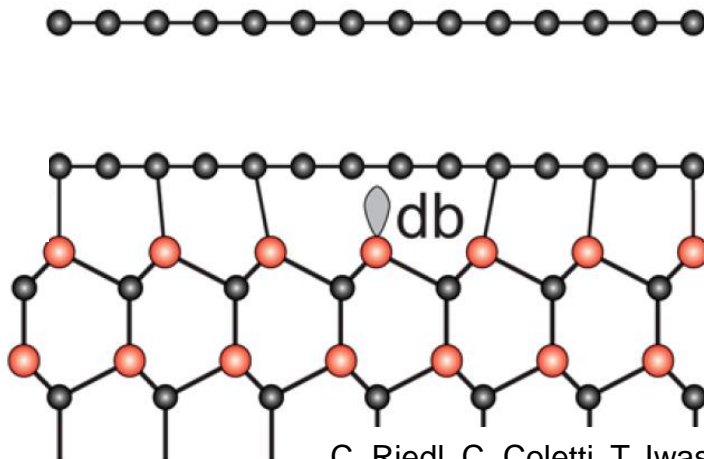


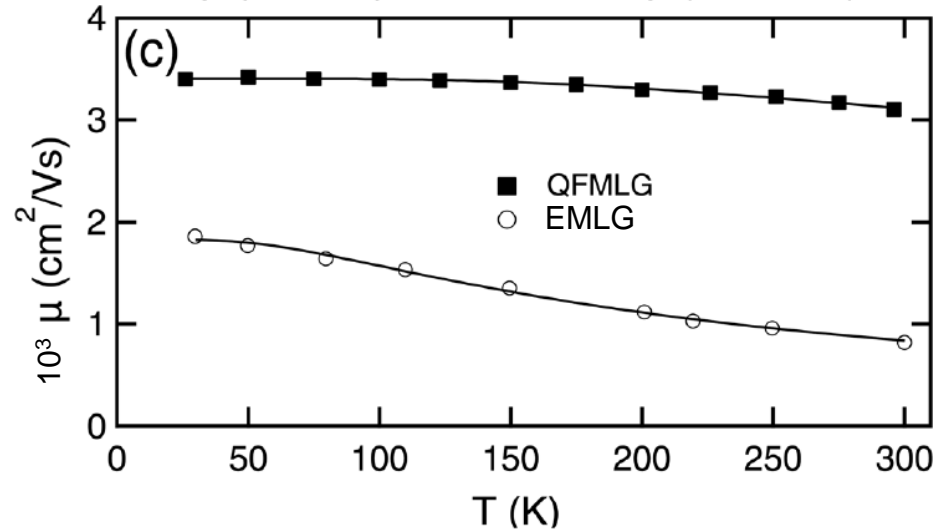
annealing

H intercalation

epitaxial monolayer
graphene (EMLG)

quasi-free-standing monolayer
graphene (QFMLG)

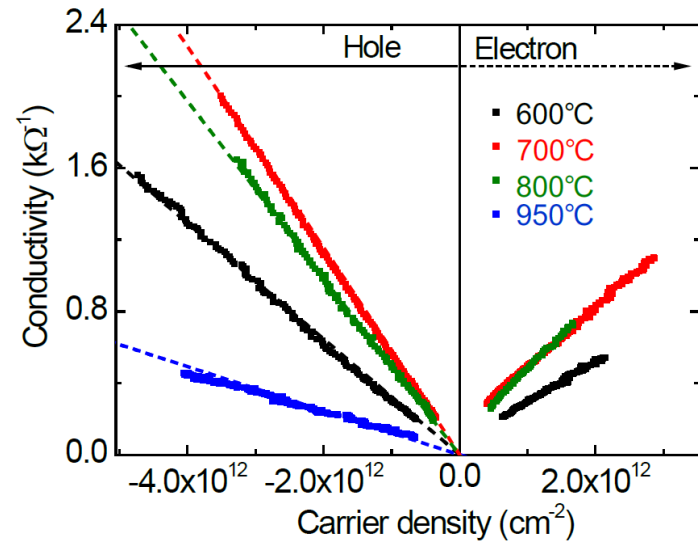
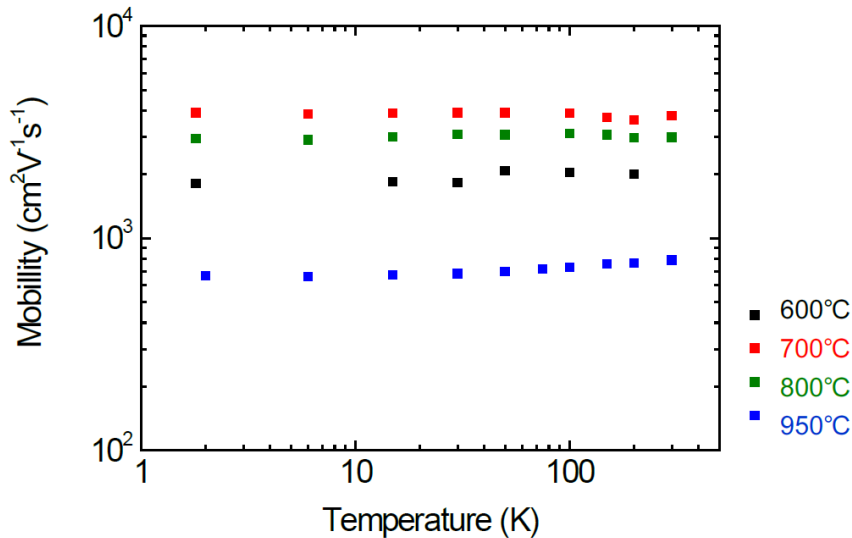




The carrier mobility of QFMLG shows less temperature dependence than EMLG, indicating less interaction between QFMLG and the SiC substrate.

However, the mobility of QFMLG ($\sim 3000 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$) is still lower than exfoliated graphene on SiO_2 or free standing graphene.

F. Speck, J. Jobst, F. Fromm, M. Ostler, D. Waldmann, M. Hundhausen, H. B. Weber, and Th. Seyller, Appl. Phys. Lett. **99**, 122106 (2011)



S. Tanabe, M. Takamura, Y. Harada, H. Kageshima, and H. Hibino, *Jpn. J. Appl. Phys.* **53**, 04EN01 (2014).

The QFMLG mobility depends on T_H ,
the substrate temperature during H intercalation

highest mobility by $T_H = 700^\circ\text{C}$

Purpose :

to observe the morphology of QFMLG formed at different T_H and
investigate the relationship with transport property

conductivity – carrier density

- linear for $T_H = 600\text{-}800^\circ\text{C}$
- charged impurity
- sublinear for $T_H = 950^\circ\text{C}$
- additional scattering by defect

Experiment

sample: 4H or 6H-SiC(0001)

cleaning anneal at 1500°C for 5 min in H₂ of 33 mbar

buffer layer growth

anneal at 1700°C for 5 min in Ar of 800 mbar

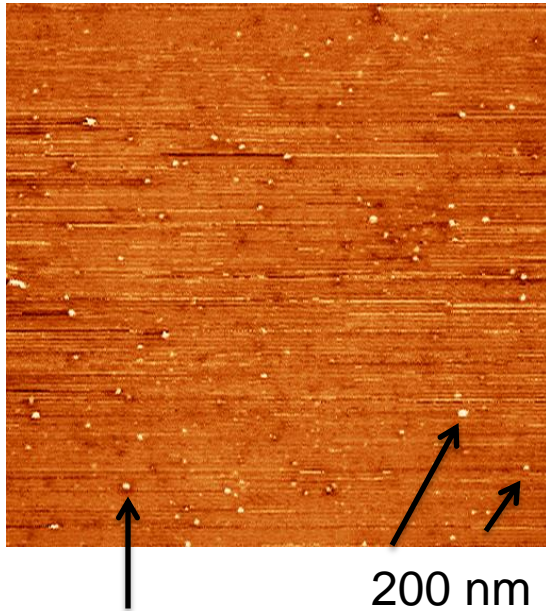
H intercalation

anneal at 600 - 1200°C for 1 hour in H₂ of 1 atm

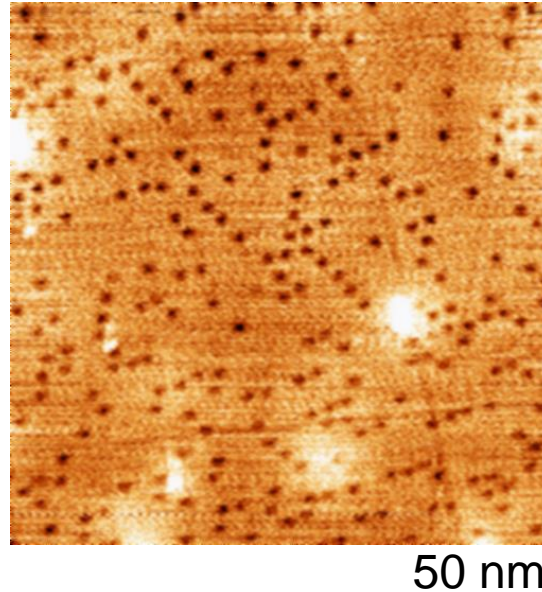
characterization

- STM in ultra-high vacuum (1×10^{-10} mbar)
- AFM in air
- TEM

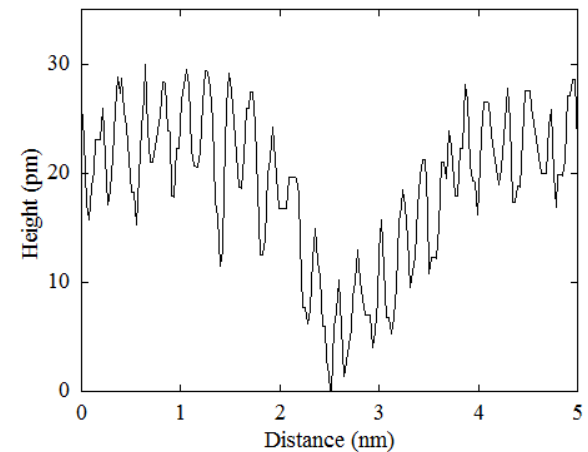
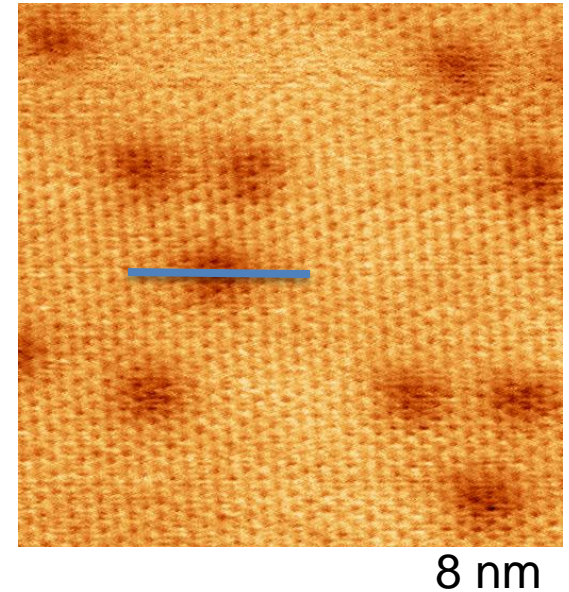
STM $T_H = 600^\circ\text{C}$



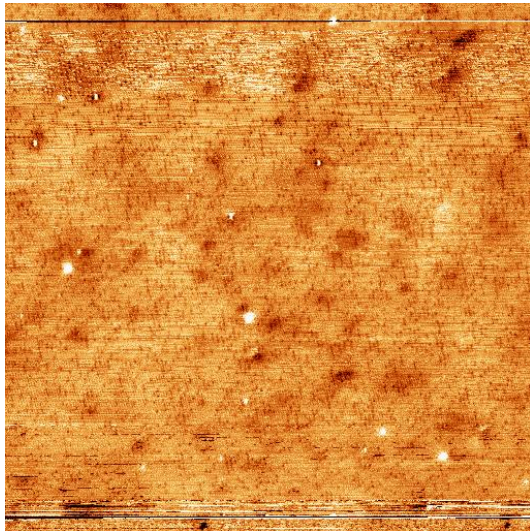
- bright spots
width: 1 nm
height: 50 pm



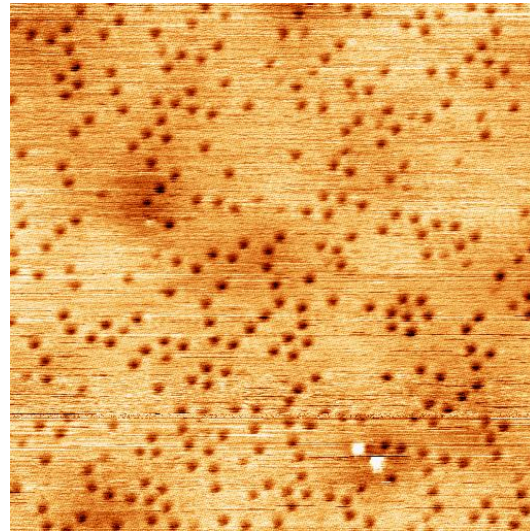
- small dark spots
width: 1.5 nm
depth: 15-25 pm



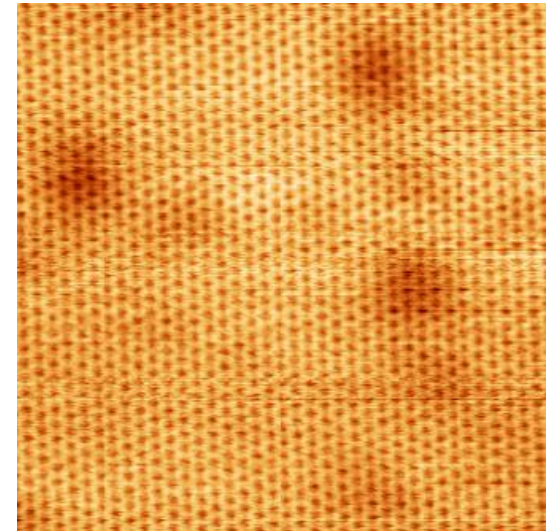
STM $T_H = 800^\circ\text{C}$



200 nm



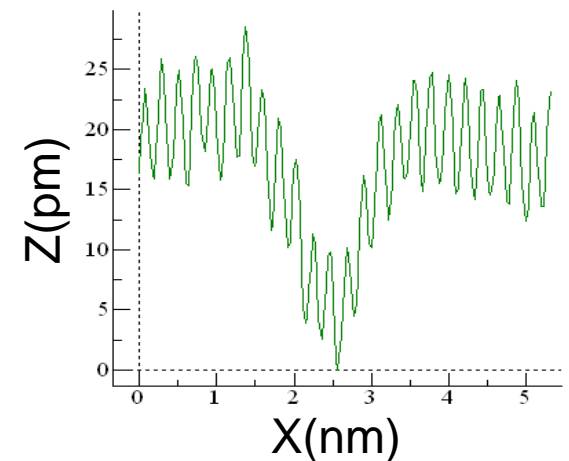
50 nm



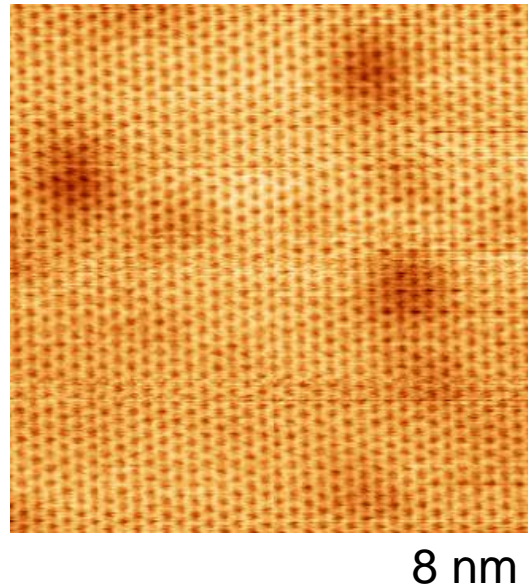
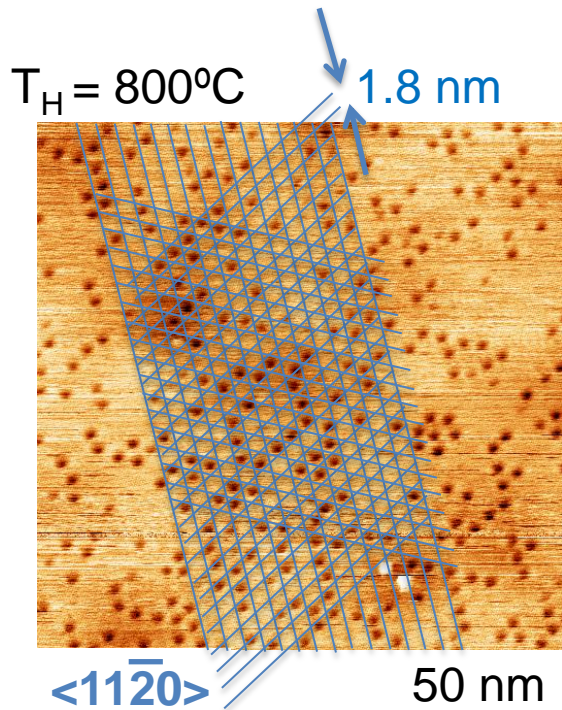
8 nm

- small dark spots
- bright spots

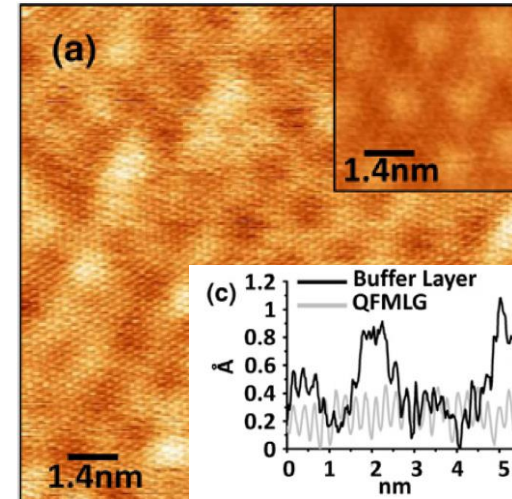
almost same morphology as $T_H = 600^\circ\text{C}$



Small dark spots



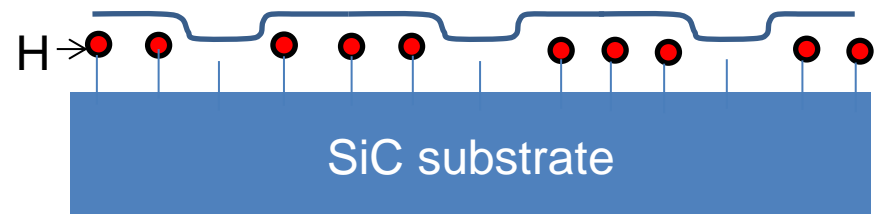
STM of a buffer layer corrugation – 60 pm



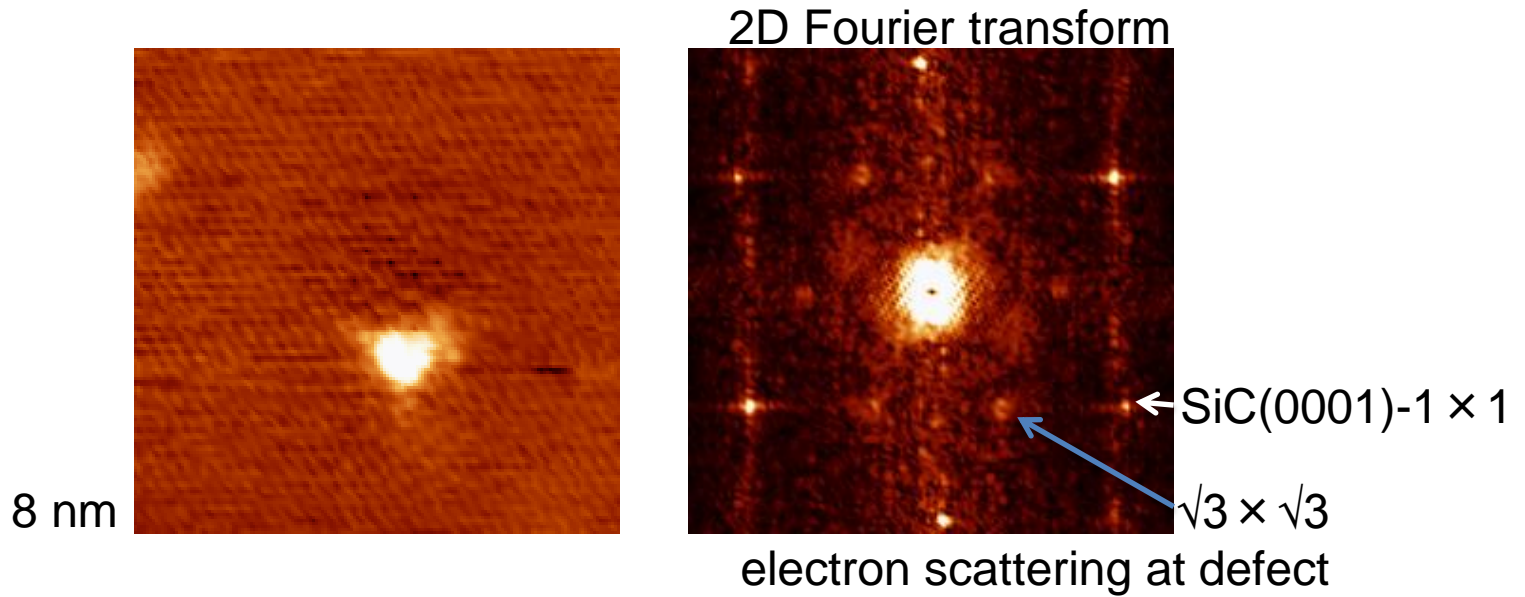
Goler, Carbon **51**, 249 (2013)

- width: 1.5 nm
- depth: 15-25 pm
- align along SiC $\langle 11\bar{2}0 \rangle$
- periodicity: 1.8 nm = SiC-6 × 6
- honeycomb inside

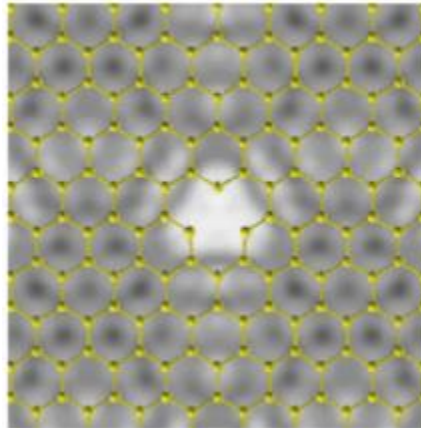
incomplete H intercalation
– Si dangling bond at interface



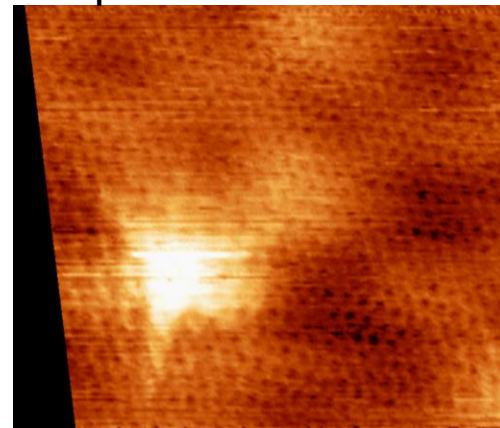
Bright spots



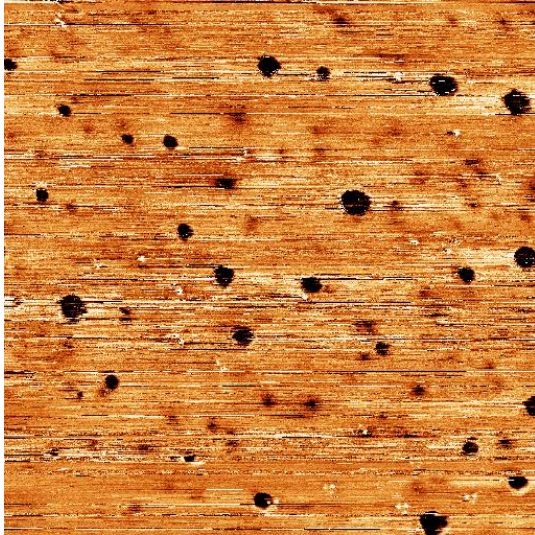
defect of EMLG



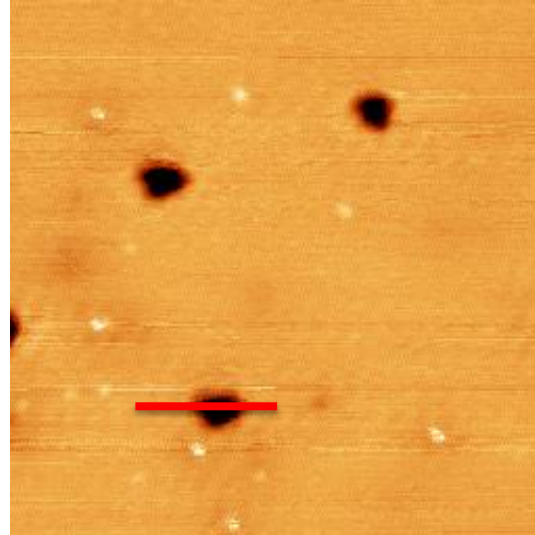
N-sputtered EMLG



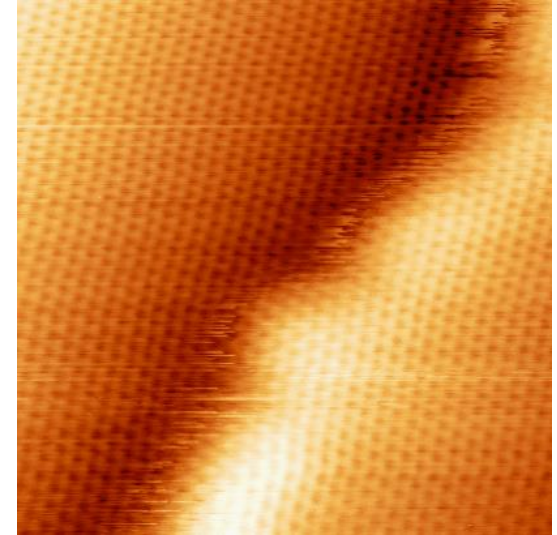
STM $T_H = 1000^\circ\text{C}$



200 nm

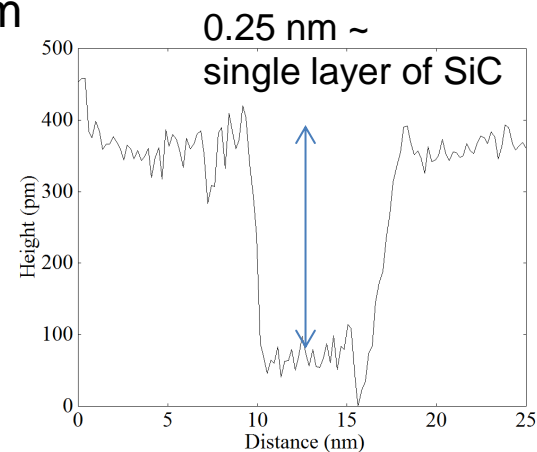


50 nm

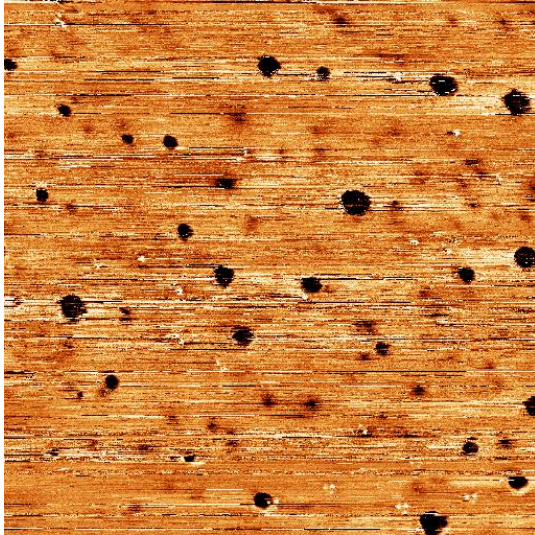


8 nm

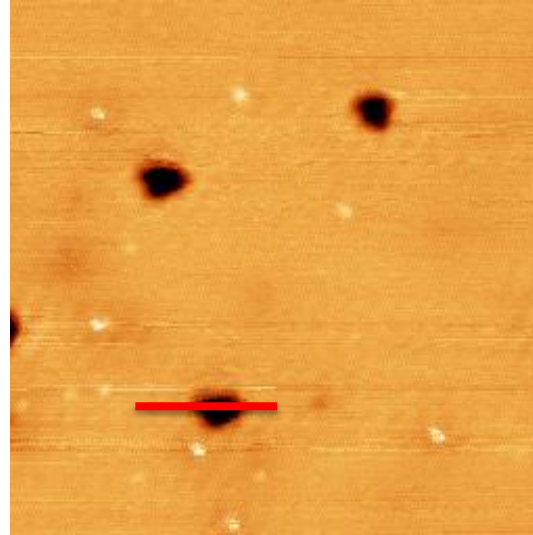
- large dark spots (width: 4-10 nm, depth: 0.25-0.3 nm)
- small dark spots (width: 1.5 nm, depth: 15-25 pm)
- bright spots



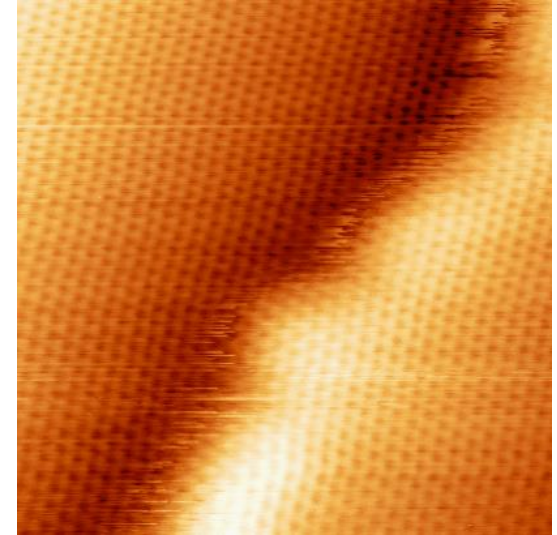
Large dark spots



200 nm

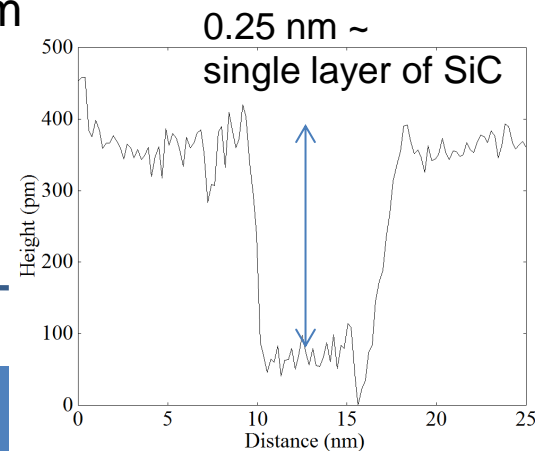


50 nm

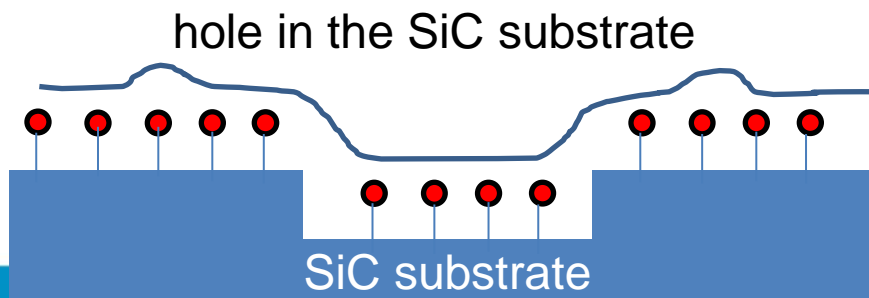


8 nm

- width: 4-10 nm, depth: 0.25-0.3 nm
- random distribution
- honeycomb inside



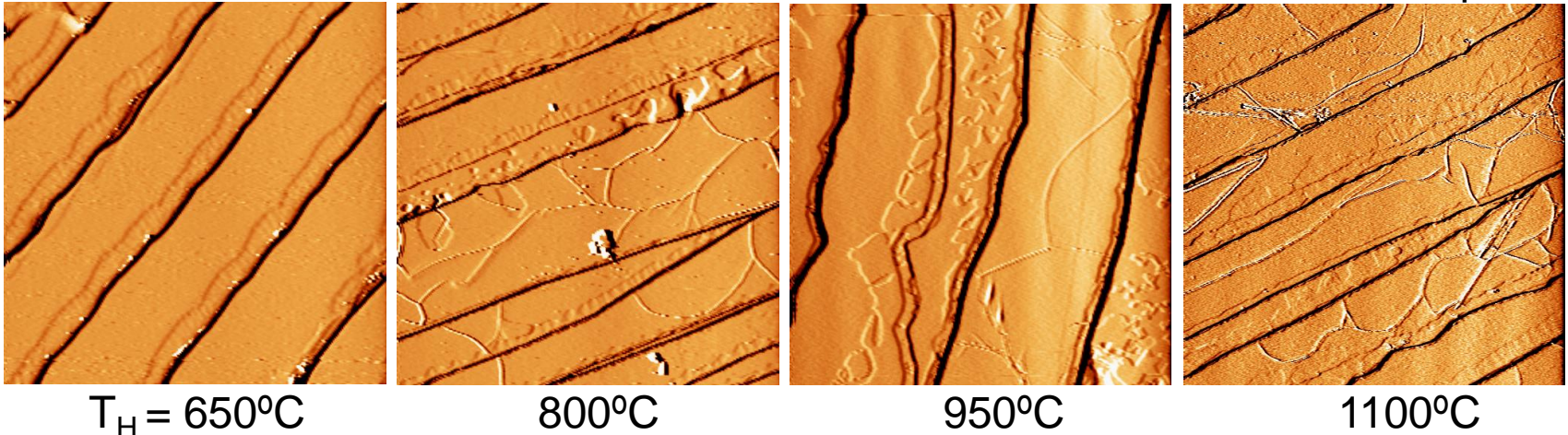
etched at high T_H



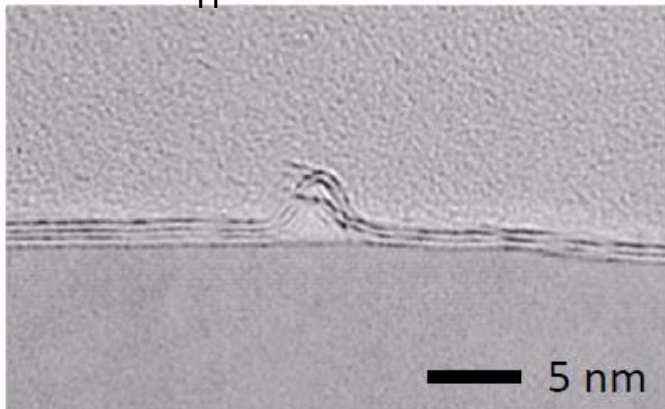
Wrinkles of graphene

AFM differentiated in horizontal axis

14 μm



TEM $T_H = 1200^\circ\text{C}$



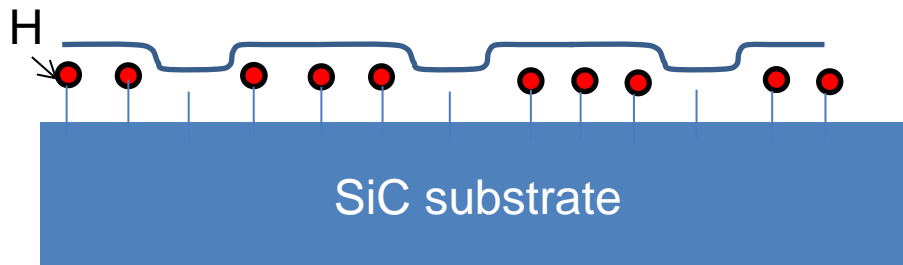
wrinkles appear at $T_H > 800^\circ\text{C}$

more frequently seen at $T_H > 1100^\circ\text{C}$

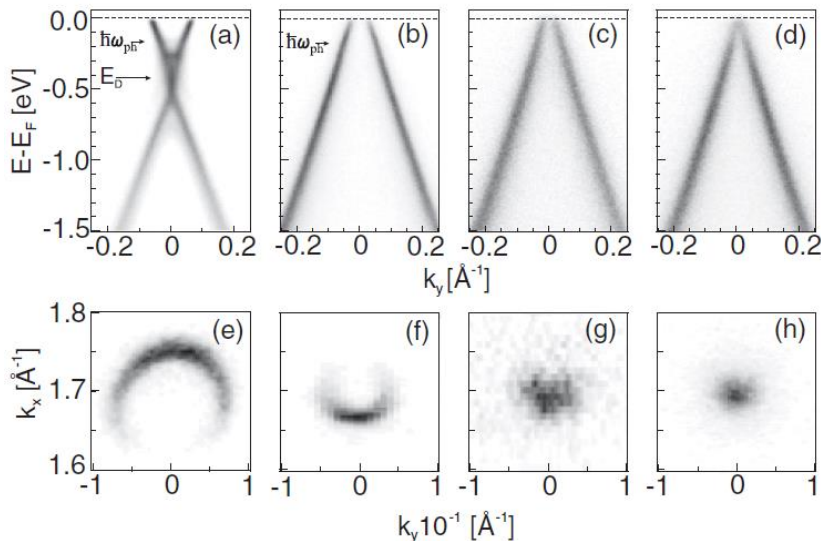
the difference in thermal expansion coefficients between graphene and SiC

Scattering in QFMLG

$T_H = 600-800^\circ\text{C}$



- small dark spots - Si dangling bonds due to incomplete H intercalation



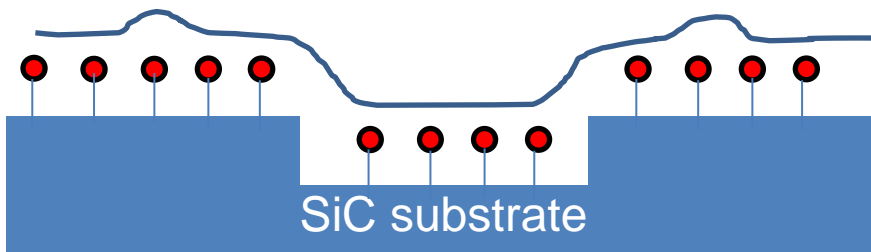
ARPES – H desorption from QFMLG

S. Forti, K. V. Emtsev, C. Coletti, A. A. Zakharov, C. Riedl, and U. Starke, Phys. Rev. B 84, 125449 (2011).

Si dangling bond donates charge to graphene and acts as a charged scattering center.

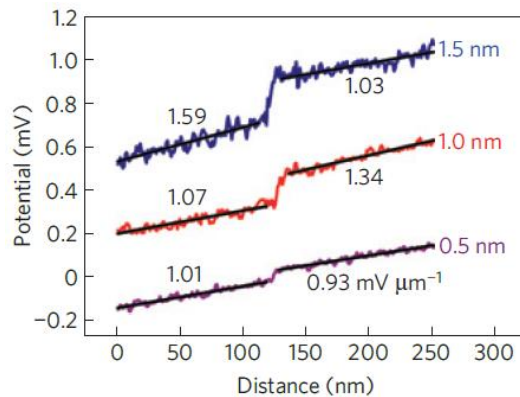
Scattering in QFMLG

$T_H = 1000^\circ\text{C}$



- dark spot – hole in SiC substrate
- wrinkle of graphene

EMLG resistance over SiC steps



Scanning tunneling potentiometry

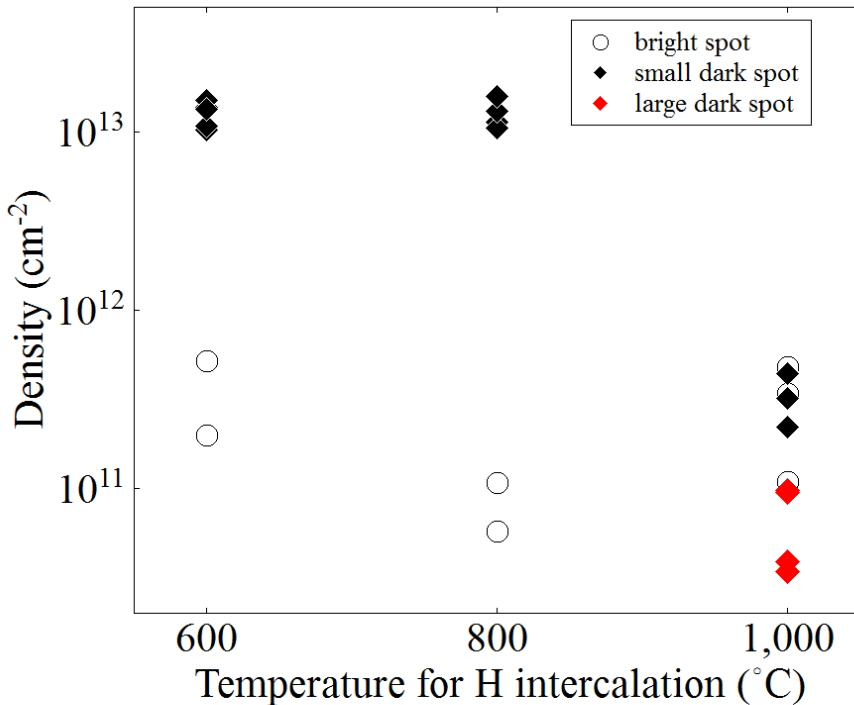
S. Ji, J. B. Hannon, R. M. Tromp, V. Perebeinos, J. Tersoff, and F. M. Ross, *Nature Materials* 11, 114 (2012)

- π - σ hybridization by curvature
- strain
- reduced doping from substrate

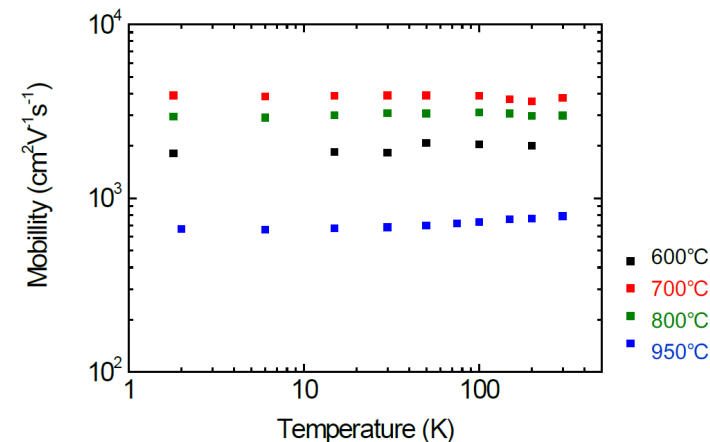
Scattering in QFMLG

As T_H increases from 600-800 to 1000°C,

- small dark spot decreases.
-more H intercalation
- large dark spot – hole in SiC substrate and wrinkles of graphene appear.
- bright spot – defect in graphene has the constant density < 0.016%.



The holes in SiC substrate and wrinkles of graphene are responsible for the lower mobility at 1000°C.



Conclusion

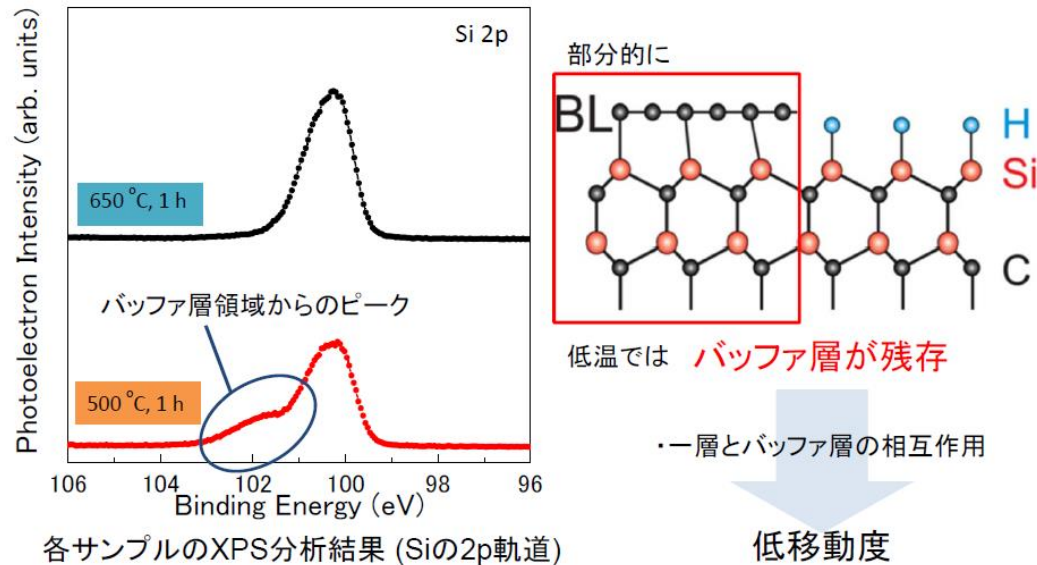
- We investigated the morphology of QFMLG formed at several temperatures by H intercalation with STM, AFM, and TEM.
- We found that Si dangling bonds due to incomplete H intercalation at the graphene-substrate interface cause carrier scattering as charged impurities in QFMLG at $T_H = 600$ and 800°C .
- At $T_H = 1000^\circ\text{C}$, holes in the SiC substrate and wrinkles of graphene appear and decrease the mobility of QFMLG, despite a better H intercalation.
- In order to obtain a higher mobility of QFMLG, we need to optimize the H intercalation condition to intercalate more H, below the temperature at which holes and wrinkles appear.

Thank you for your attention!

Funding:

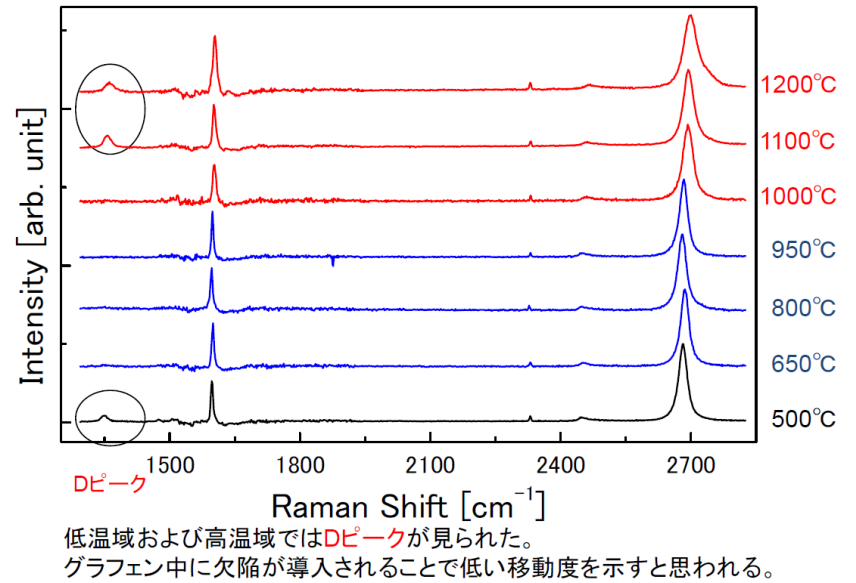
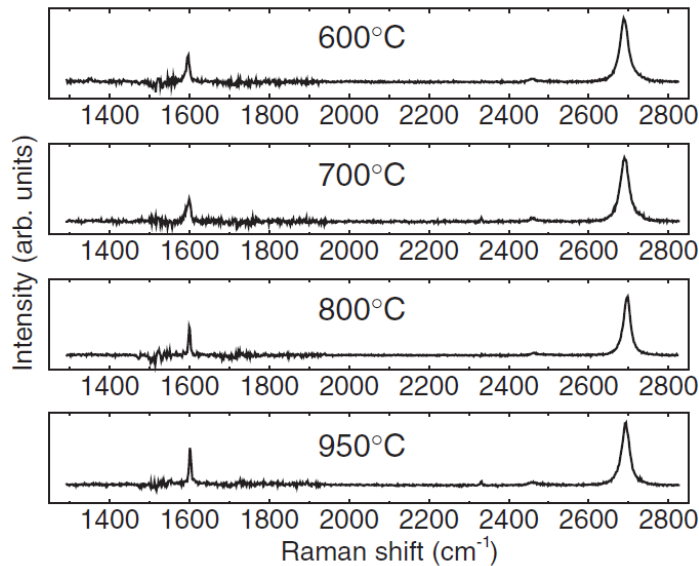


XPS



The signal from buffer layer was obtained at T = 500C, but not at 650C.

Raman



D peak at 600C, 500C and > 1100C