# **Growth and characterization of graphene** on SiC(0001) and SiC(000-1)

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# Graphene on SiC(0001)

- Zerolayer (ZL), monolayer (ML) and bilayer (BL) graphene •Buffer layer of C atoms arranged in a graphene-like honeycomb structure and
- covalently bound to Si atoms.
- •Better graphene thickness control and uniformity.
- •Defined azimuthal orientation with respect to the substrate. •Ordered stacking of layers..

# **ATOMIC FORCE MICROSCOPY (AFM)**

### H-etched Si-face



Hydrogen etching removes polishing damages and creates atomically flat terraces



# Graphene on SiC(000-1)

- Few layers (FL) graphene
- •Difficult control of the number of layers during growth
- •Different azimuthal orientations.
- •Electronically decoupled graphene layers (turbostatic stacking of graphene layers).
- •Higher mobilities.

# **ATOMIC FORCE MICROSCOPY (AFM)**

## H-etched C-face



Half or full unit cell high steps after hydrogen etching cover the surface sample.







Si face epi-ready SiC substrate

10 µm

#### Epitaxial Graphene growth on Si-face



Tapping mode AFM topography and phase images showing graphene monolayer (lighter contrast) and bilayer (darker contrast) domains.

#### **RAMAN SPECTROSCOPY**



2 3 4 μm

1,0 1,5 2,0 2,5

C face polishing scratches on SiC substrate

21 nm

-4 nm

### Epitaxial Graphene growth on C-face



- Tapping mode AFM topography of two different regions, showing domains separated by narrow ridges (2-3 nm high), step bunching, and wider terraces.

#### **RAMAN SPECTROSCOPY**





μm

0,5

-1.0







#### SCANNING TUNNELLING MICROSCOPY (STM)



The STM image reveals the typical (6v3x6v3)R30° and the (6x6) superstructures which are indicated by the dashed and the solid diamond, respectively.

ig. from S. **Goler et al**. **Carbon**, 51 249, (**2013**).

The honeycomb lattice signature of graphene is visible on top of the long range superstructure due to the interaction of the ZL with the substrate.



# Effect of HF treatment on graphene on C face



• Raman spectra of graphene measured at 532 nm, after the subtraction of the SiC background signal, show G peak at ~1590 cm<sup>-1</sup> and 2D peak at  $\sim$ 2700 cm<sup>-1</sup>. • 2D peak fit with one Lorentian, FWHM 34

cm<sup>-1</sup>.

• 2D/G map shows a high ratio between 0.7 and 2.5 with low values at the terraces edge.

#### SCANNING TUNNELLING MICROSCOPY (STM)



The absence of the buffer layer on the C face causes a rotational disorder due to stacking faults.

The Moiré patterns visible in (A) and (B) are due to misorientation bewteen the two outermost layers.

The periodicity D of the pattern is given by D=a/[2] $sin(\vartheta/2)$ , where *a* is the lattice parameter (*a* is the graphite atomic lattice constant, 0.246 nm) and  $\vartheta$  is the misorientation angle between the layers. From the measured values of D we derive  $\vartheta_A = 4.8^\circ$  and  $\vartheta_B = 1.6^\circ$ 

> There is also an area (C) without Moiré pattern, that probably corresponds to a AB (Bernal) stacking or to small periodicities that are difficult to resolve.

No HF



#### 10- **No HF** 10] **HF** 200 400 600 200 400 600

Energy (eV)

AUGER ELECTRON SPECTROSCOPY (AES)

#### No HF-treated sample

• >50% corrugated "worm-like" area in the large scale STM.

• on a smaller scale STM reveals a highly corrugated surface, graphene is however continuous and defect free. • O Auger signal visible.

#### • the LEED pattern lacks of sharpness.

#### LOW ENERGY ELECTRON DIFFRACTION (LEED)



#### HF-treated sample

• the surface is flat as revealed by larger and smaller scale STM images. • oxygen completely removed by HF as confirmed by Auger analysis. • typical LEED patter of FL graphene on C-face with diffraction spots indicating several preferred orientations.

Energy (eV)

-10

-15