

# Hydrogen Vacancies in Quasi-Free-Standing Monolayer Graphene

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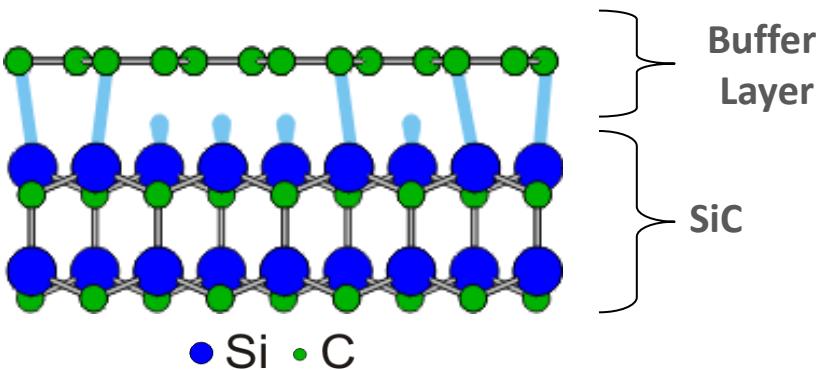
*3 IBM Zurich Research Laboratory, Zurich, Switzerland*

*4 NTT Basic Research Laboratories, NTT Corporation, Japan*

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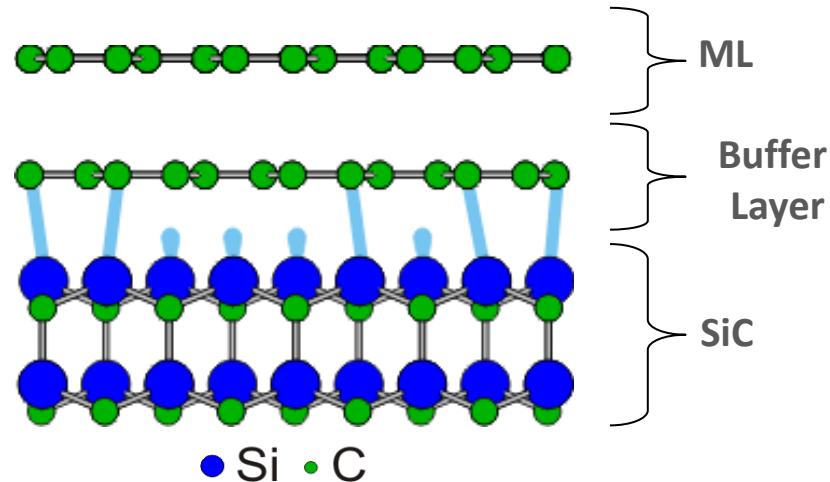
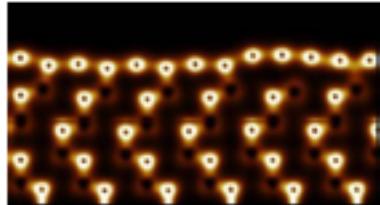


# Graphene growth on SiC(0001)

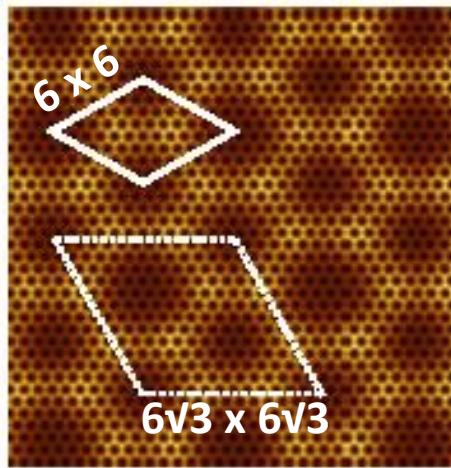


**Buffer Layer**

Topologically identical atomic carbon structure as graphene.  
Does not have the electronic band structure of graphene due to periodic  $sp^3$  C-Si bonds.



Theoretical Calculations



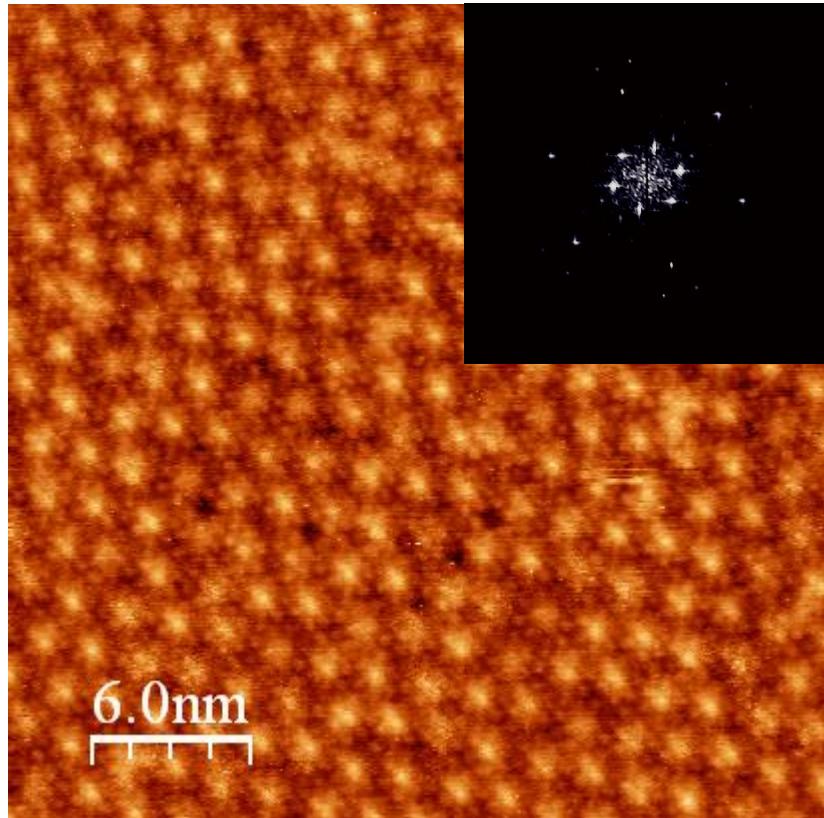
Superstructure of both the buffer layer and monolayer graphene on the Si face from the periodic interaction with the substrate.

F. Varchon, et al., PRB 77, 235412 (2008).

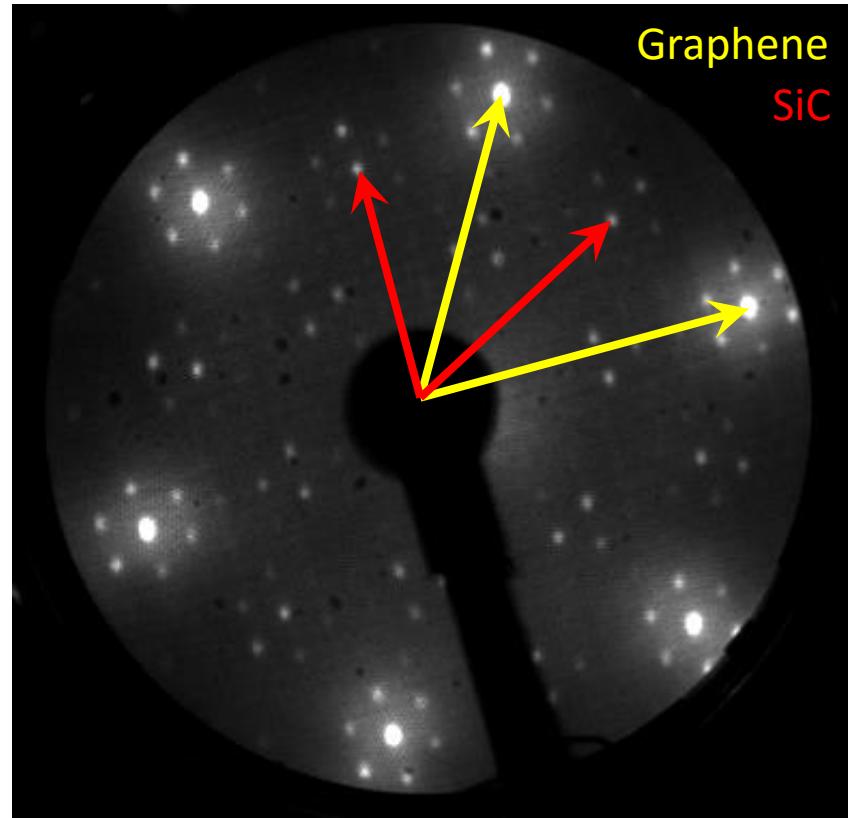
F. Varchon, et al., PRB 77, 235412 (2008).

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# Superstructure

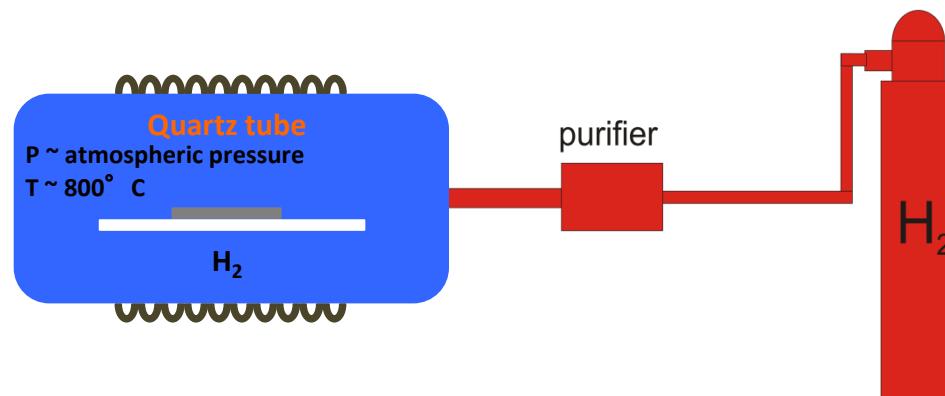
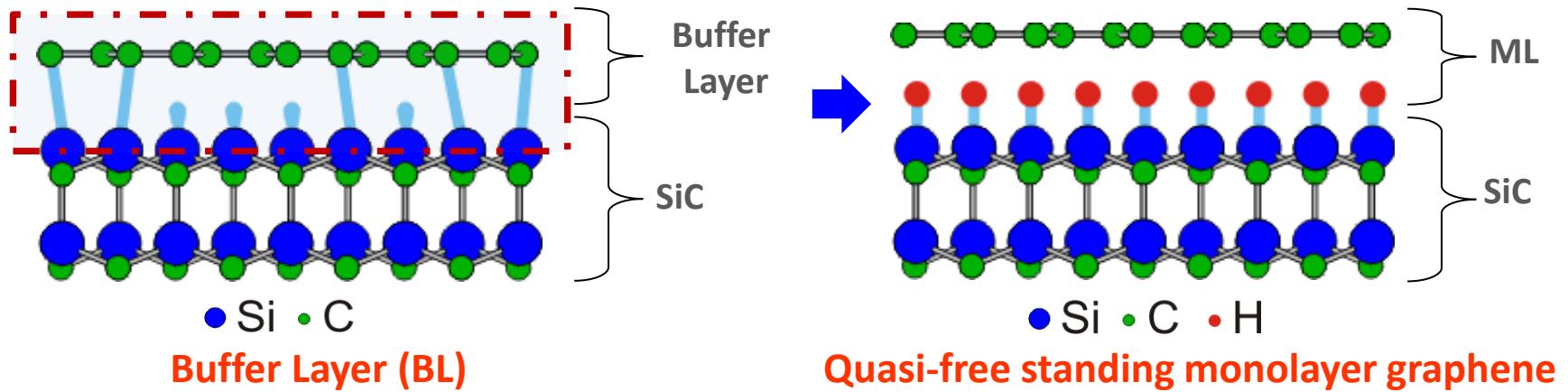


30 nm, 1V, 100 pA



$E = 75\text{ eV}$

# Hydrogen Intercalation

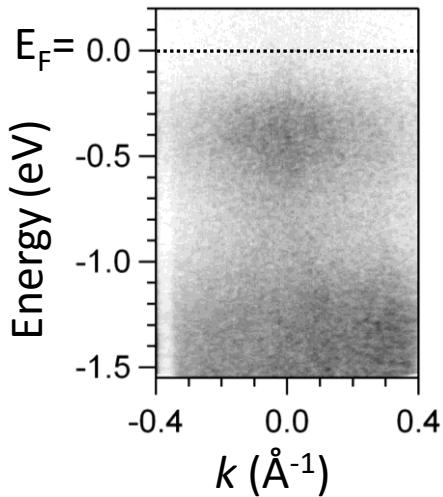
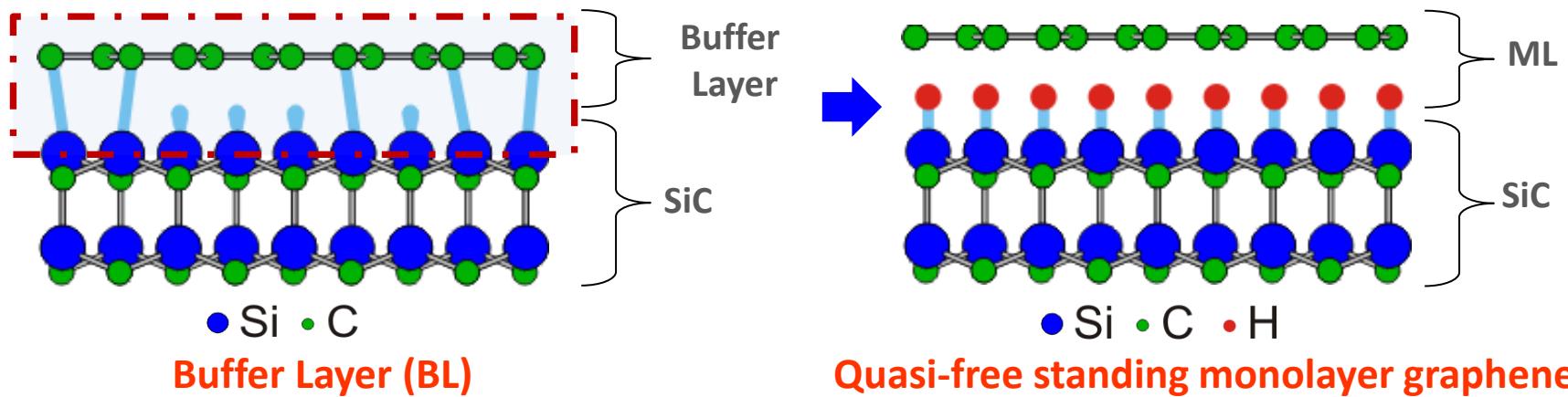


C. Riedl, C. Coletti, T. Iwasaki, A. A. Zakharov, and U. Starke,  
 Phys. Rev. Lett. **103**, 246804 (2009)

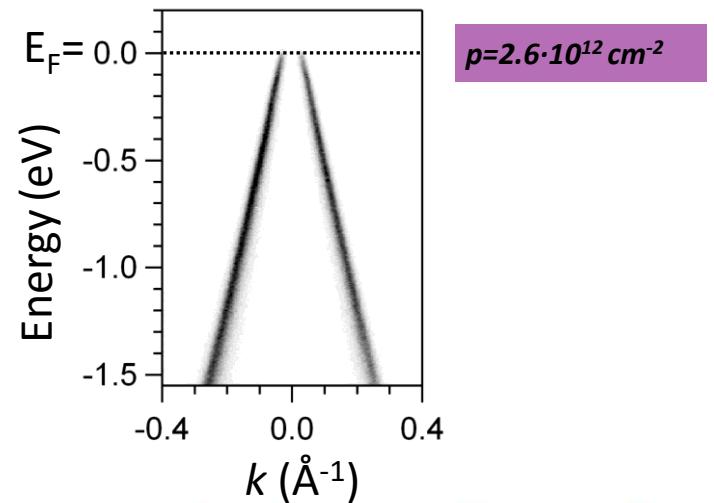
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# Hydrogen Intercalation



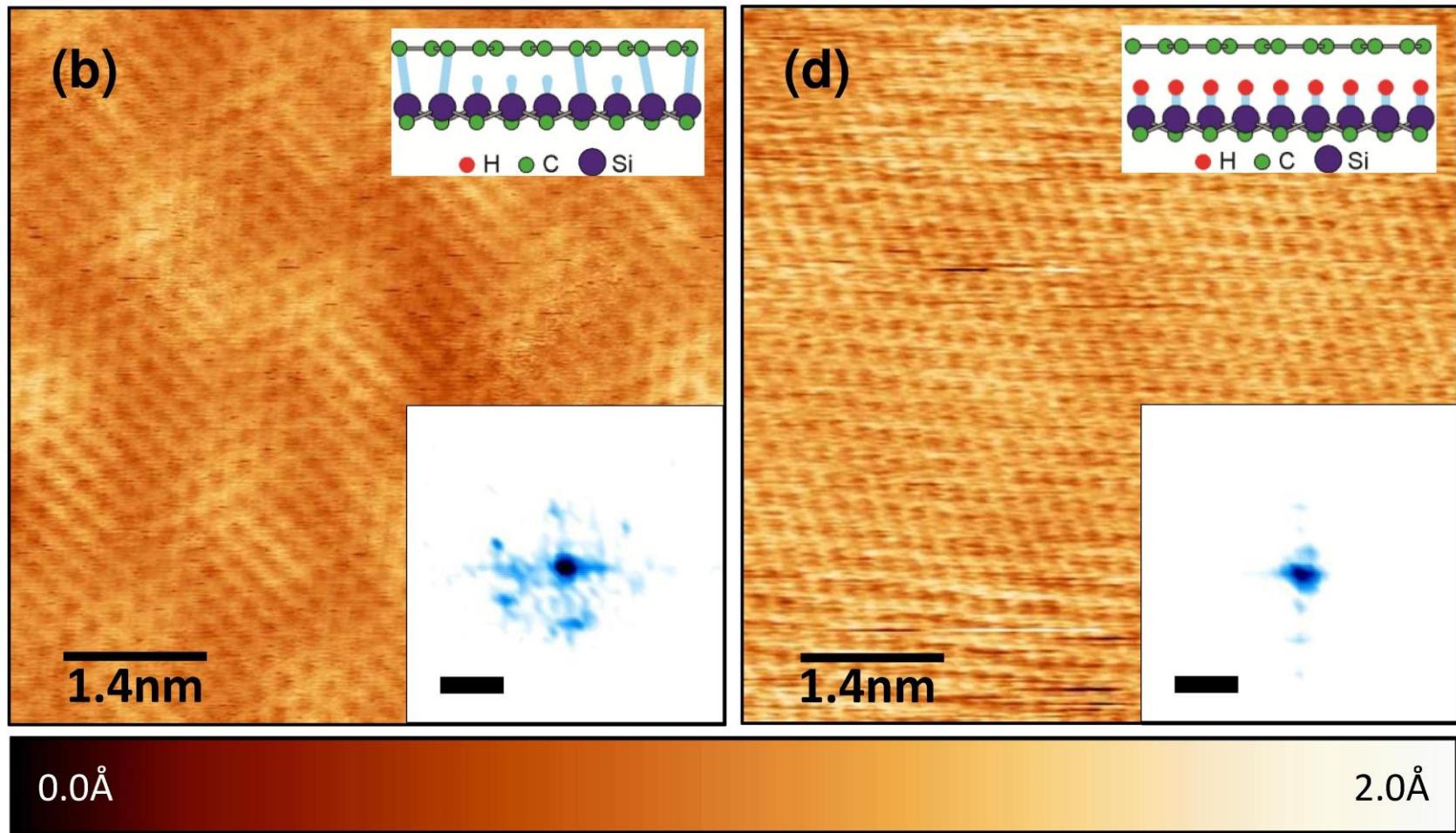
Quasi-free standing monolayer graphene



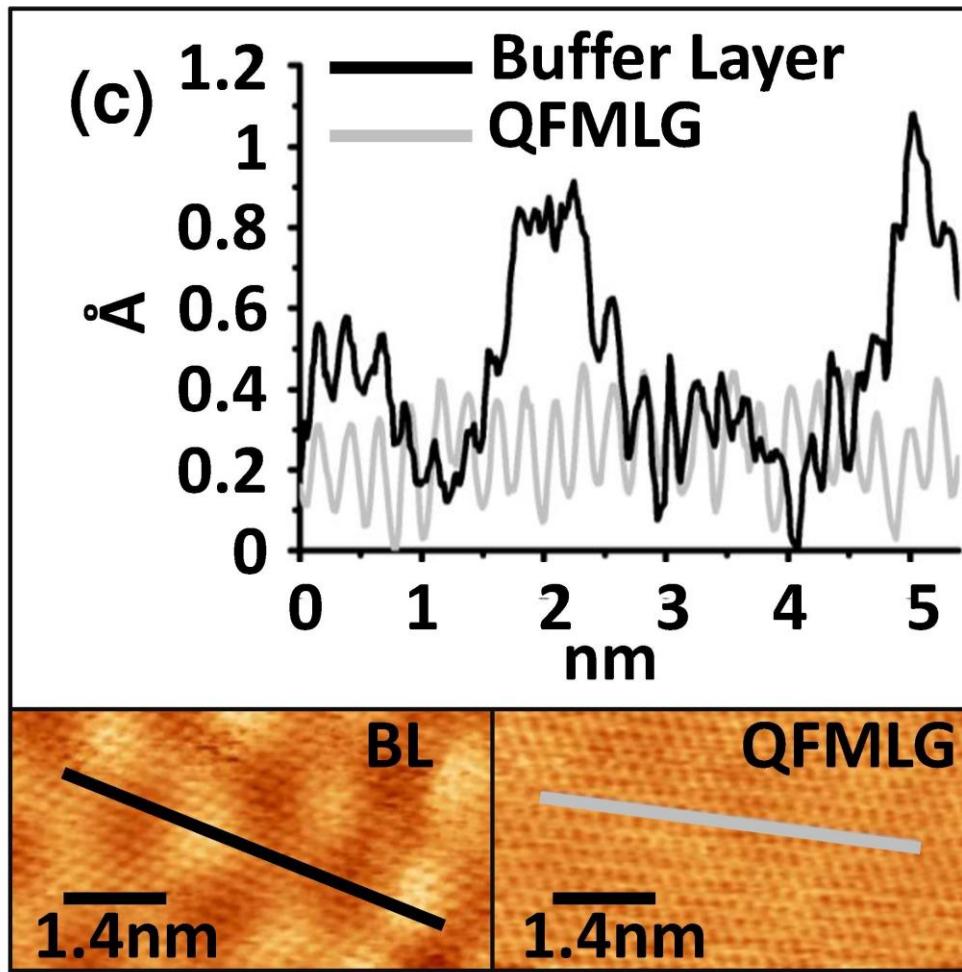
C. Riedl, C. Coletti, T. Iwasaki, A. A. Zakharov, and U. Starke,  
Phys. Rev. Lett. **103**, 246804 (2009)

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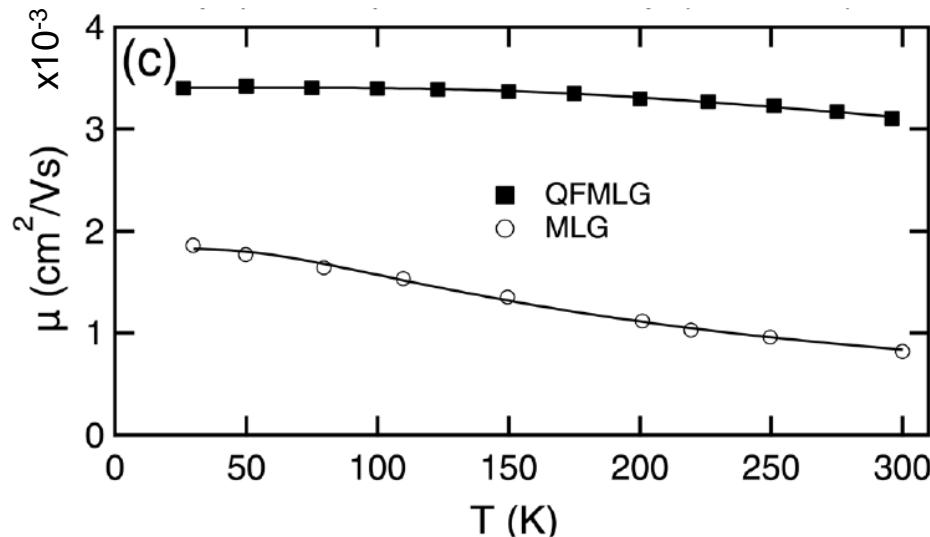
# BL vs. QFMLG



# BL vs. QFMLG



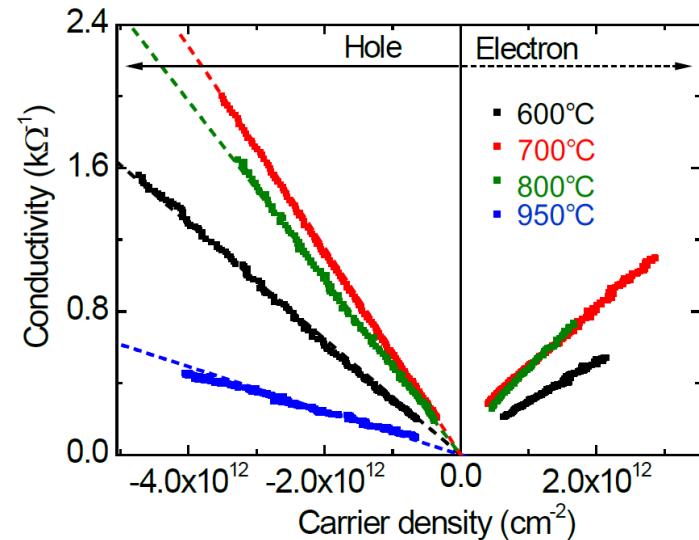
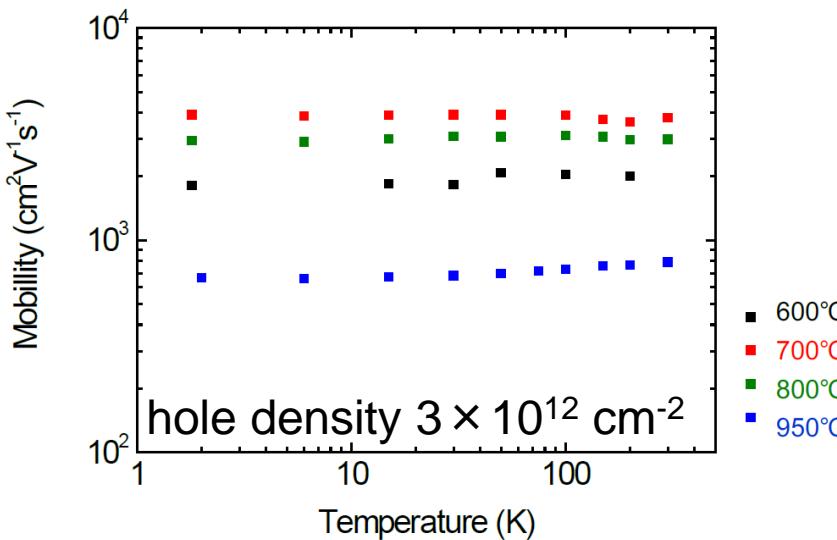
# Carrier Mobility



F. Speck, J. Jobst, F. Fromm, M. Ostler, D. Waldmann, M. Hundhausen, H. B. Weber, and Th. Seyller, Appl. Phys. Lett. **99**, 122106 (2011).

The carrier mobility of QFMLG shows less temperature dependence than MLG, indicating less interaction between QFMLG and the SiC substrate.

However, the mobility of QFMLG is still lower than that of exfoliated graphene on  $\text{SiO}_2$  or free standing graphene.



S. Tanabe, M. Takamura, Y. Harada, H. Kageshima, and H. Hibino, Jpn. J. Appl. Phys. **53**, 04EN01 (2014).

The QFMLG mobility depends on  $T_H$ ,  
the substrate temperature during H intercalation

Highest mobility at  $T_H = 700^\circ\text{C}$

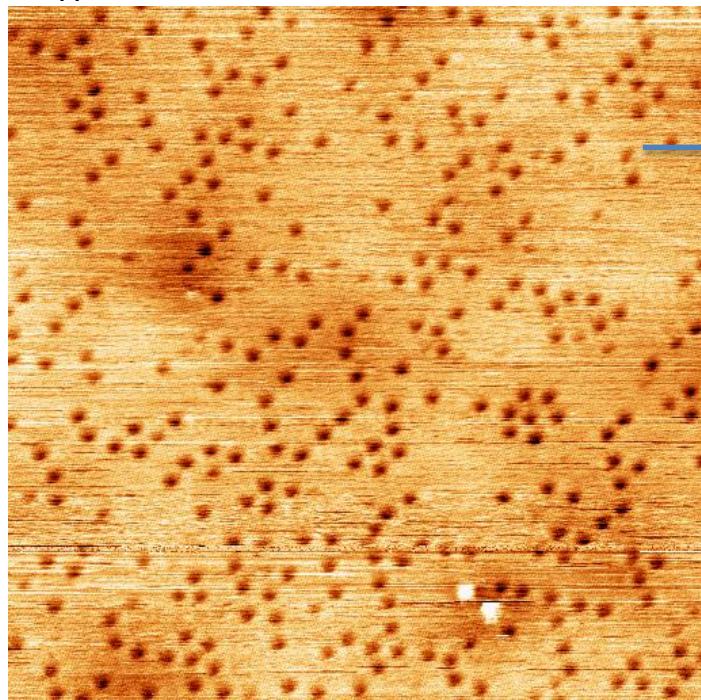
Our Purpose :  
to observe the morphology of QFMLG formed at different  $T_H$  and  
investigate the relationship with transport property

conductivity vs. carrier density:

- linear for  $T_H = 600-800^\circ\text{C}$   
- charged impurities
- sublinear for  $T_H = 950^\circ\text{C}$   
- additional scattering by defects

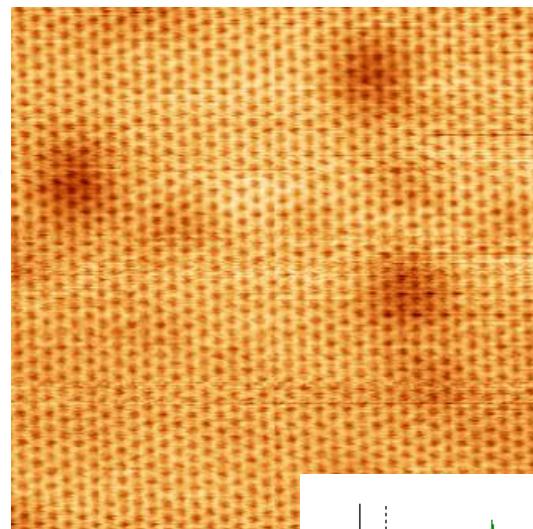
# Intercalation at 600-800°C

$T_H = 800^\circ\text{C}$  sample

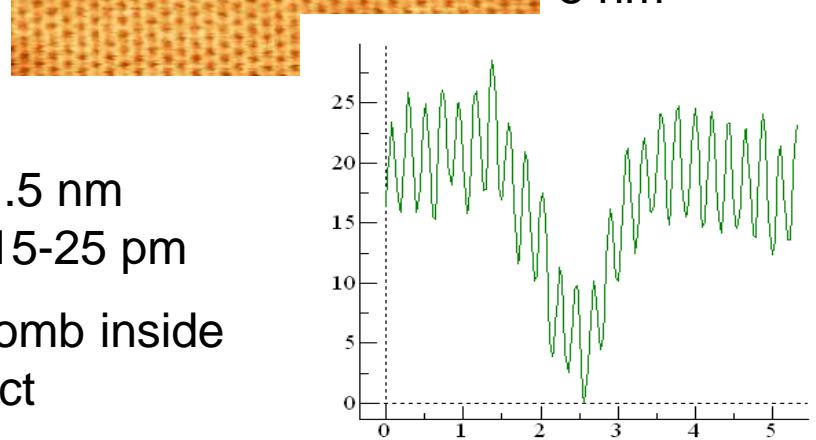


50 nm  
0.6 V, 0.4 nA

- dark spot width: 1.5 nm



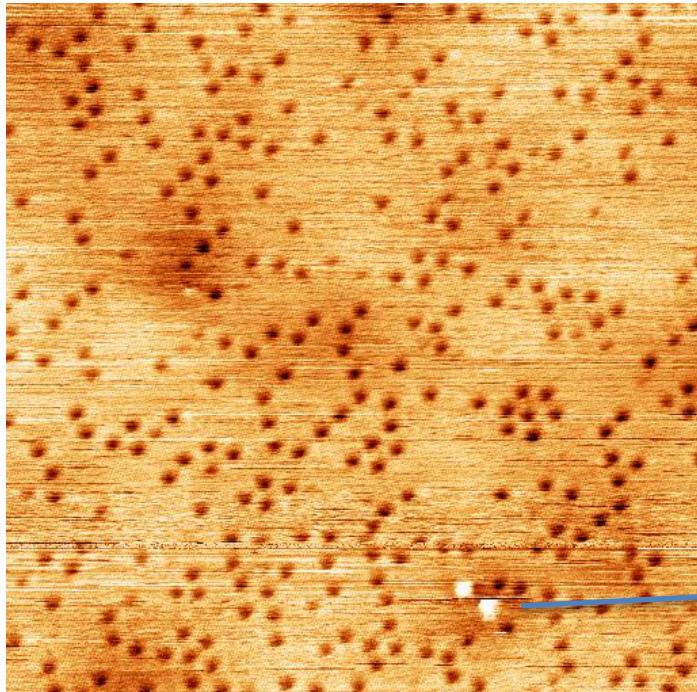
- width: 1.5 nm
- depth: 15-25 pm
- honeycomb inside
- no defect



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# Intercalation at 600-800°C

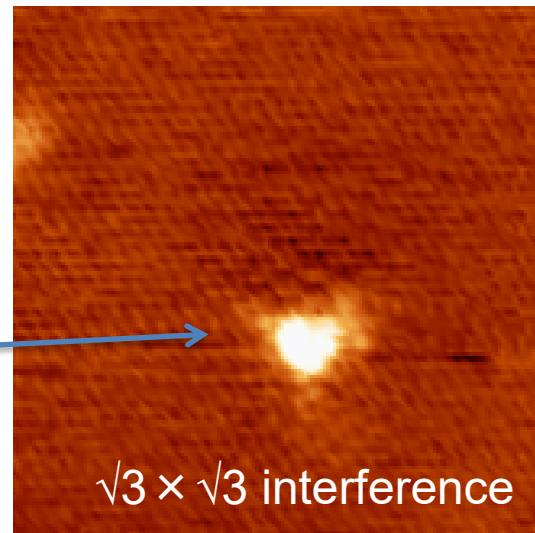
$T_H = 800^\circ\text{C}$  sample



50 nm

0.6 V, 0.4 nA

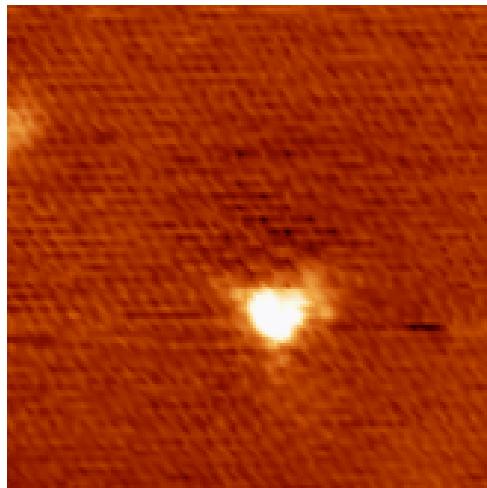
- graphene defect



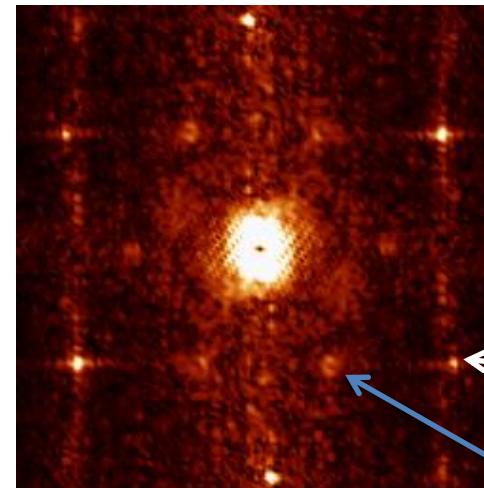
8 nm  
0.8 V, 0.4 nA

# Graphene defect

8 nm  
0.8 V, 0.4 nA



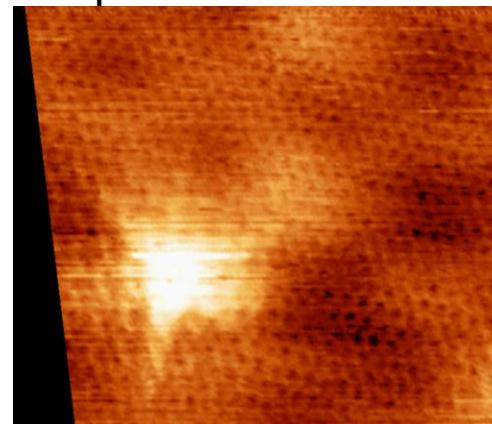
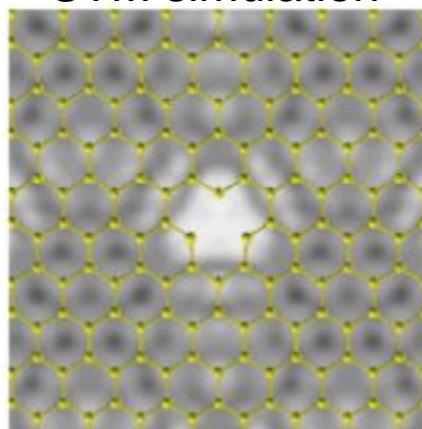
2D FFT



graphene- $1 \times 1$   
 $\sqrt{3} \times \sqrt{3}$

electron interference at defect

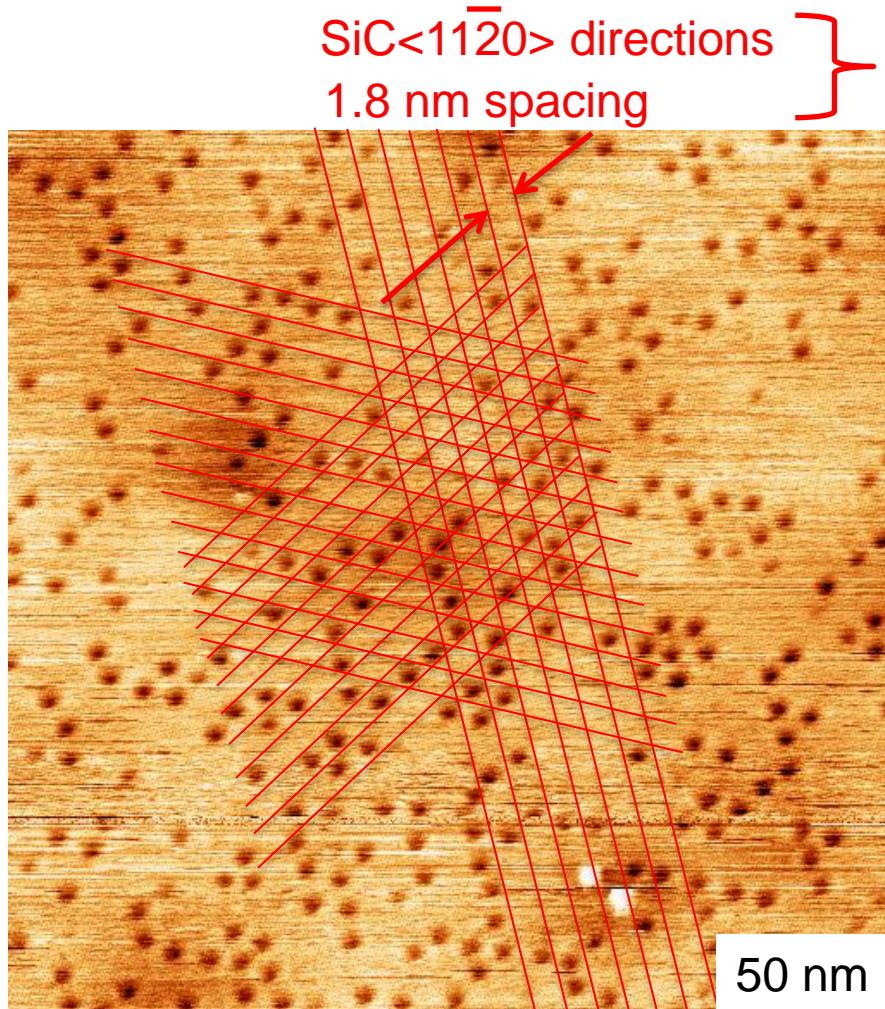
STM simulation  
N-sputtered EMLG



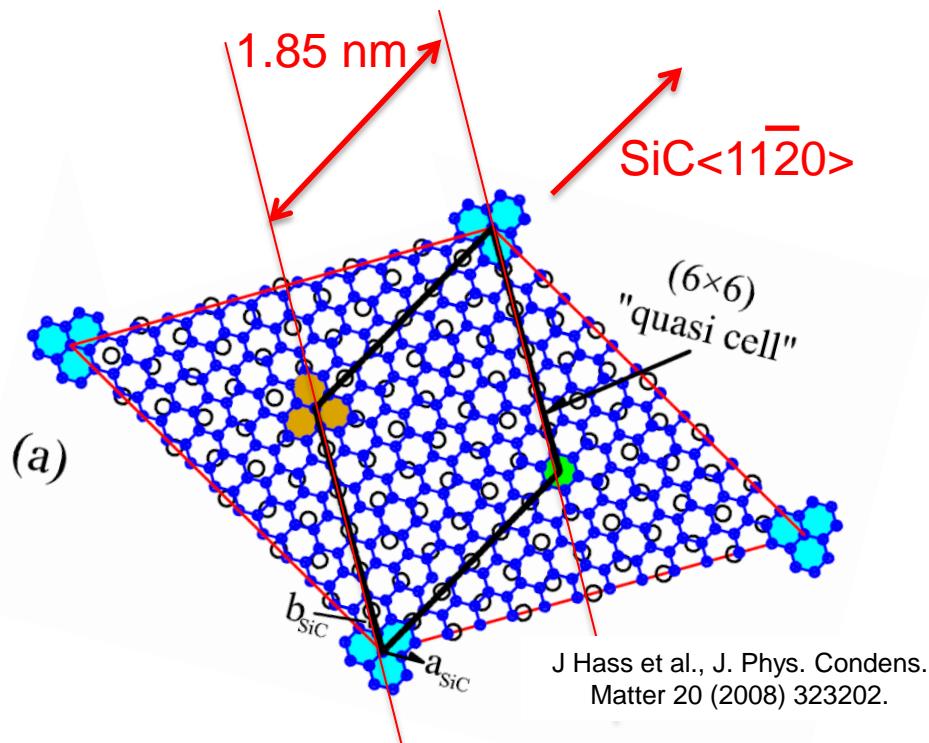
S. H. Rhim, Appl. Phys.  
Lett. **100**, 233119 (2012).

T. Mashoff et al., Appl. Phys.  
Lett. **106**, 083901 (2015).

# Intercalation at 600-800°C

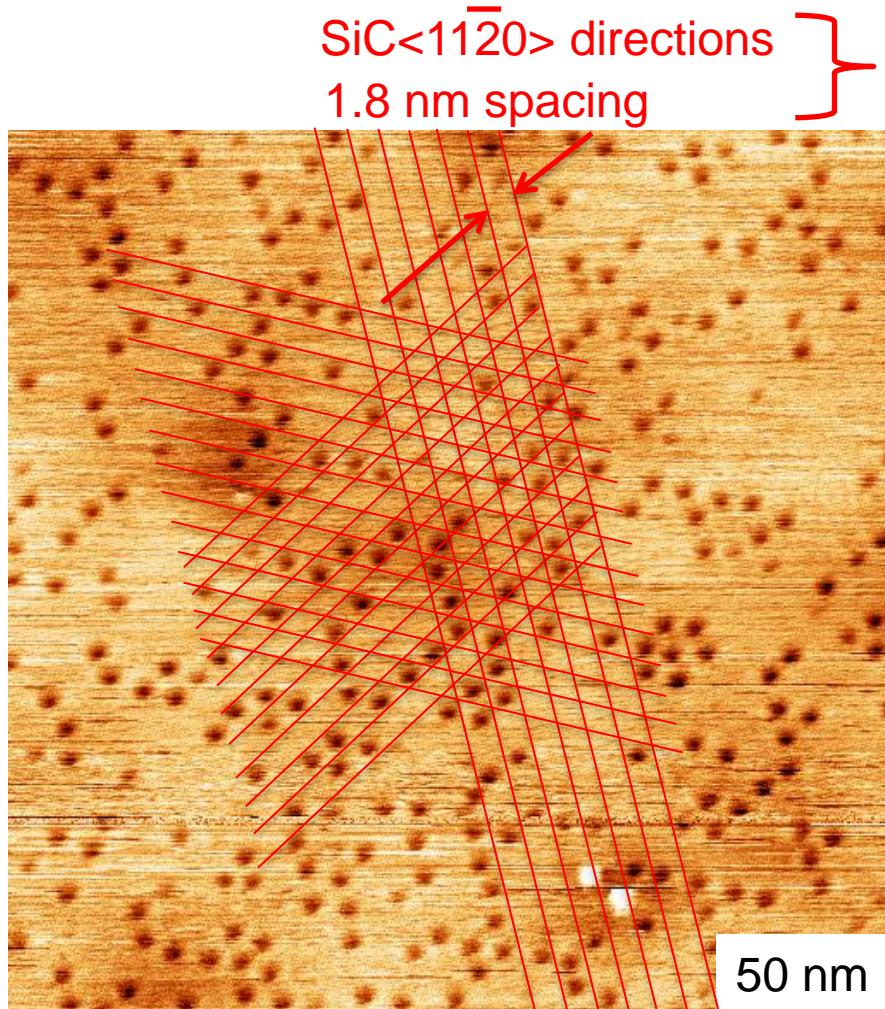


~ SiC 6 × 6 cell  
~ quasi cell of Moiré pattern produced by graphene and SiC(0001) lattices



blue: graphene  $a = 0.24589$  nm  
white:  $a = Si 0.30805$  nm

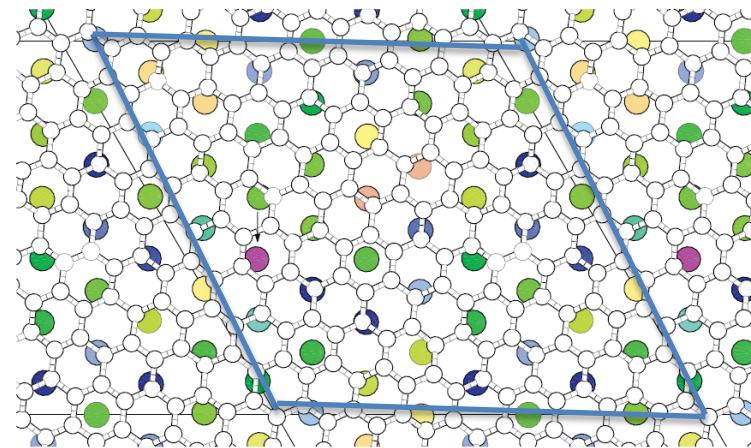
# Hydrogen Vacancies



~ SiC  $6 \times 6$  cell

~ quasi cell of Moiré pattern produced by graphene and SiC(0001) lattices

spatial distribution of hydrogenation energy on Si sites in  $4\sqrt{3} \times 4\sqrt{3}$  buffer layer



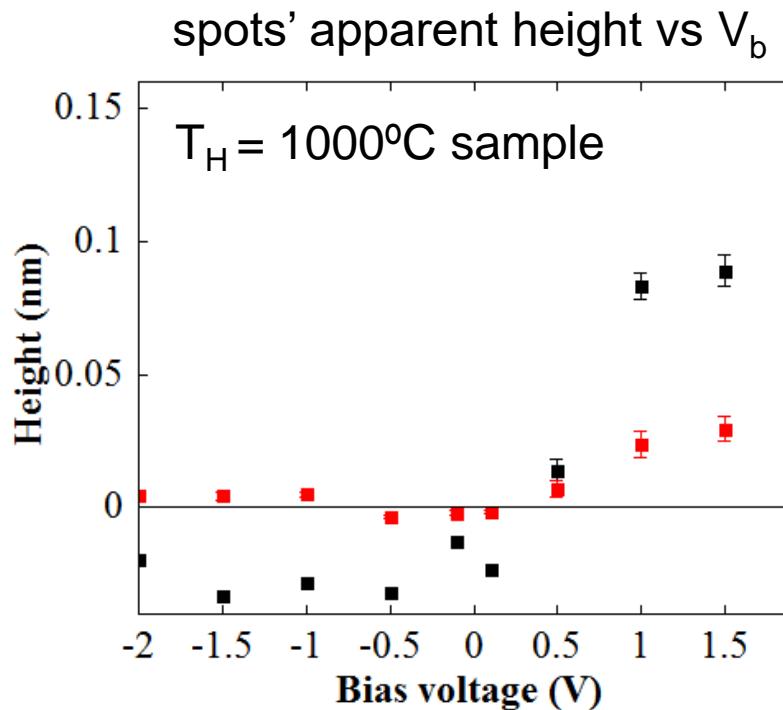
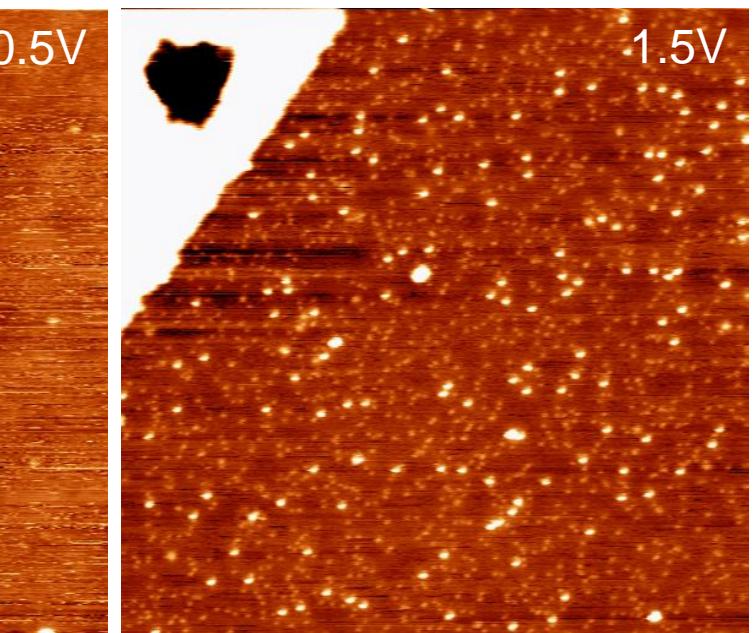
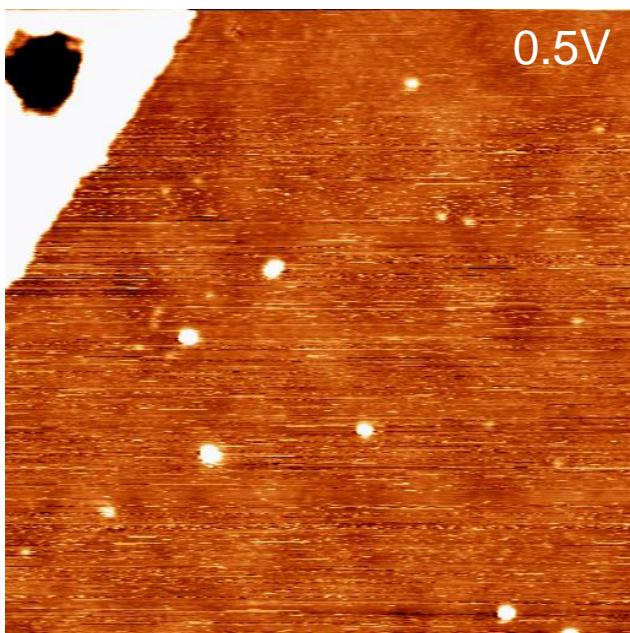
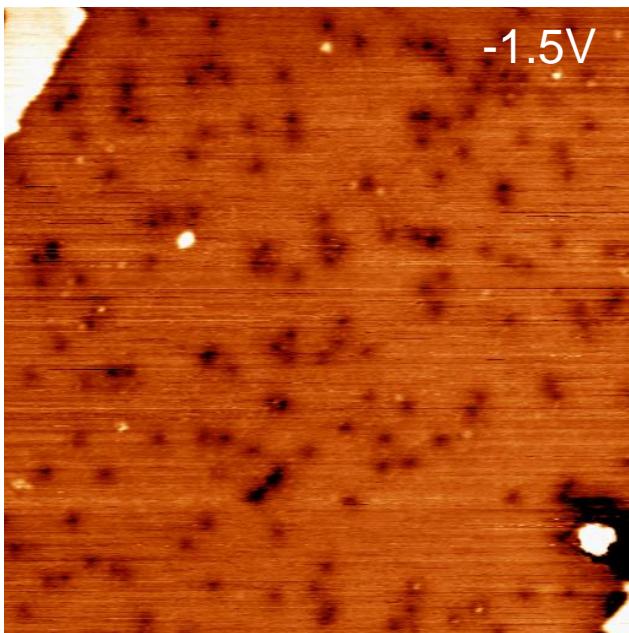
dark blue: the least favored H adsorption site

G. Sclauzero, A. Pasquarelo, Appl. Surf. Sci. 291, 64 (2014).

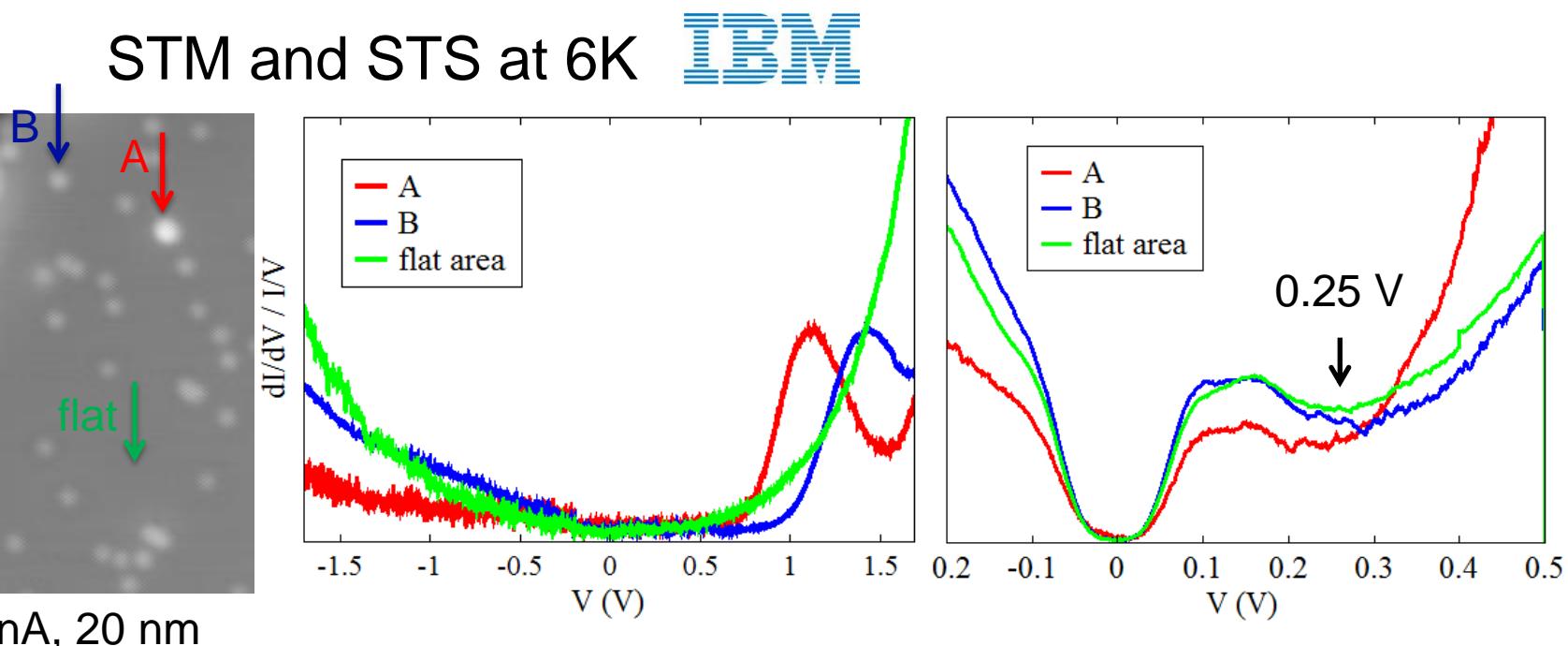
Si dangling bonds distribute along the periodicity of the Moiré pattern

- spots' apparent height varies with  $V_b$
- electronic effect rather morphology
- two types of spots

0.1nA 100nm



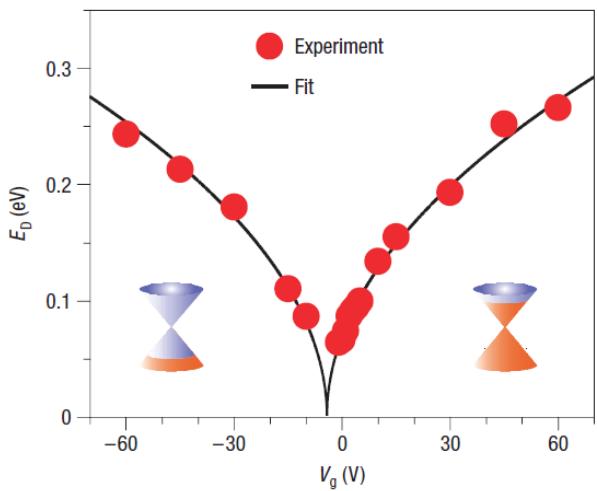
# Si dangling bonds



gap like feature at 0 V  
dip at 0.25 V

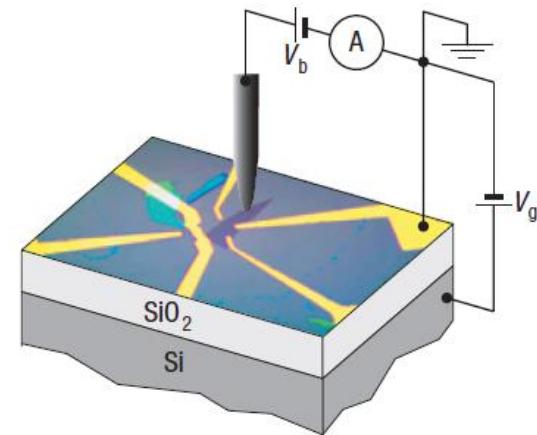
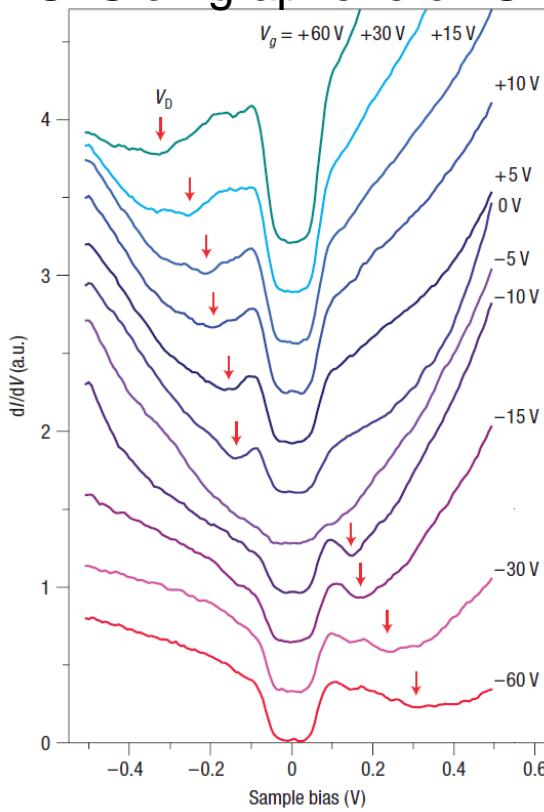
# Si dangling bonds

Energy position of Dirac point vs Vg



$$E_D = \hbar v_F \sqrt{\pi \alpha |V_g - V_0|}$$

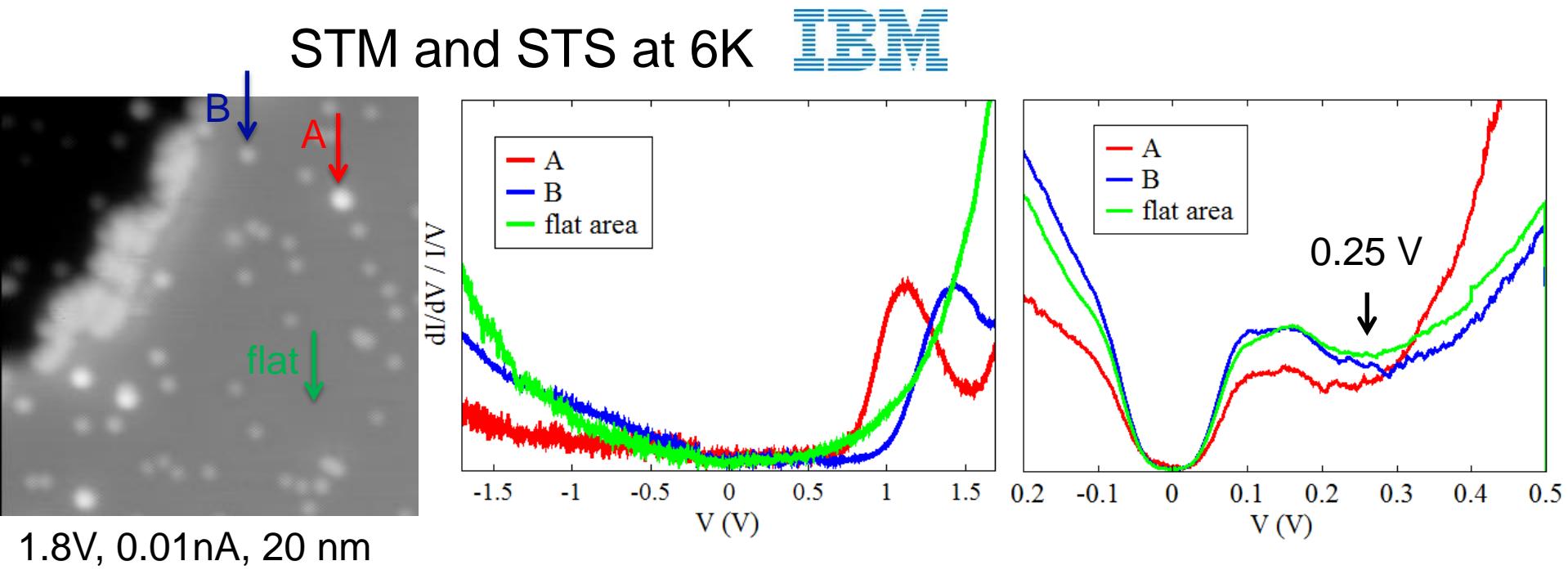
STS on graphene on  $\text{SiO}_2$  with  $V_g$



Zhang, et.al,  
Nature Phys. 4, 627 (2008).

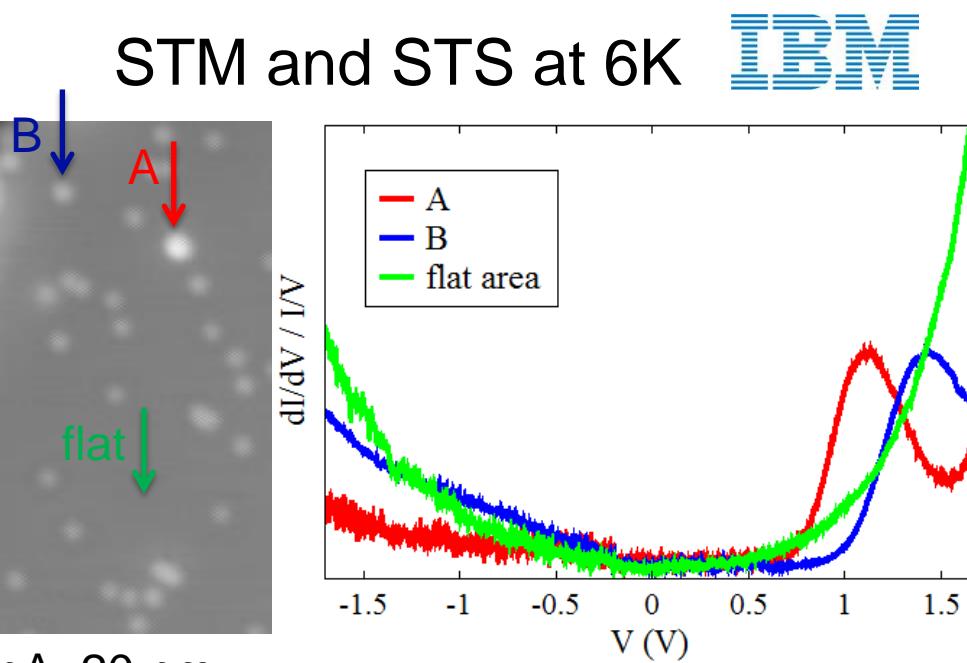
- The dip moves with  $V_g$ , corresponds to energy position of Dirac point
- Gap like feature at 0V due to suppression of tunneling to states with large momentum and tunneling enhancement at higher energy due to a phonon-mediated inelastic channel ( $\pm 63\text{meV}$ )

# Si dangling bonds

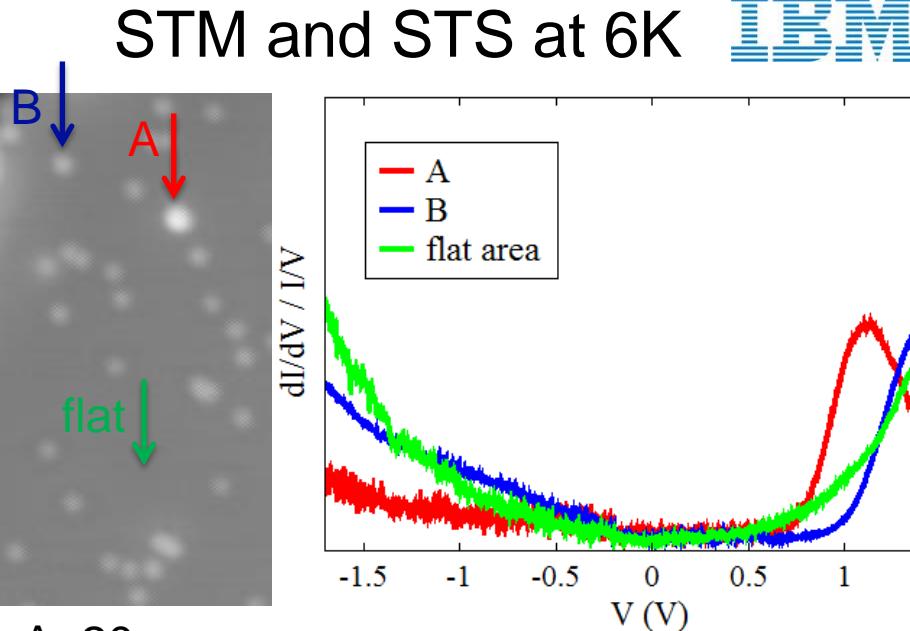


- p-type doping of QFMLG
- consistent with other experimental (ARPES) and theoretical reports  
 C. Riedl et al., Phys. Rev. Lett. **103**, 246804 (2009).  
 J. Sławińska et al., Carbon 93, 88 (2015).
- spontaneous polarization of polar surface on SiC substrate  
 J. Ristein et al., PRL 108, 246104 (2012).

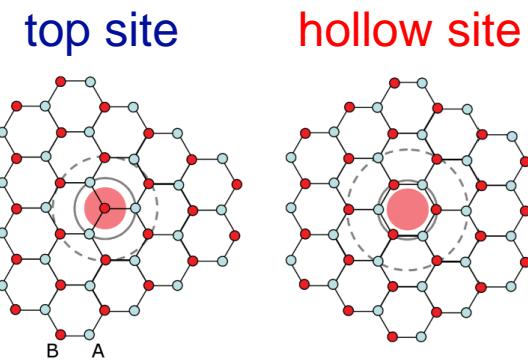
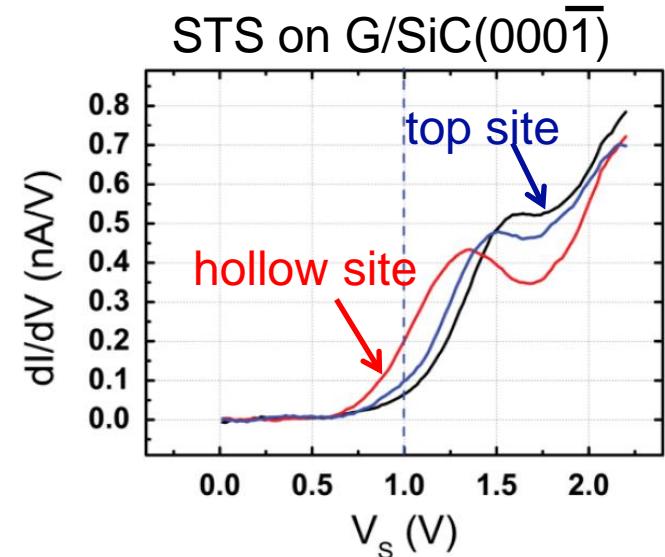
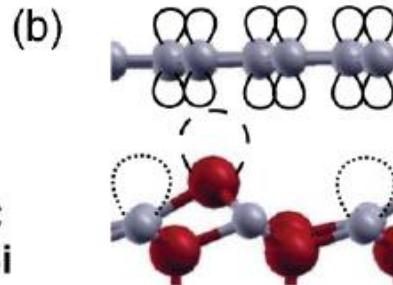
# Si dangling bonds



# Si dangling bonds



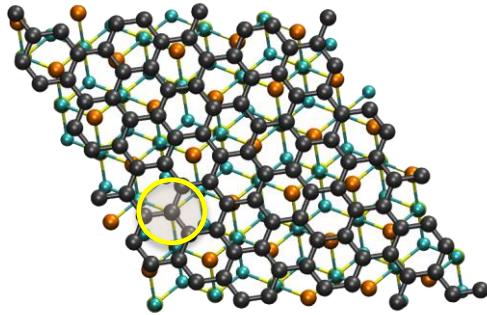
2 types of Si dangling bonds at different graphene / Si stacking configurations



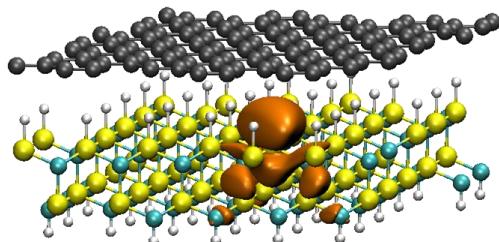
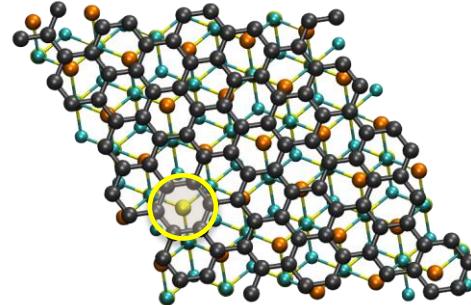
# DFT: Single H vacancy

## Single H vacancy

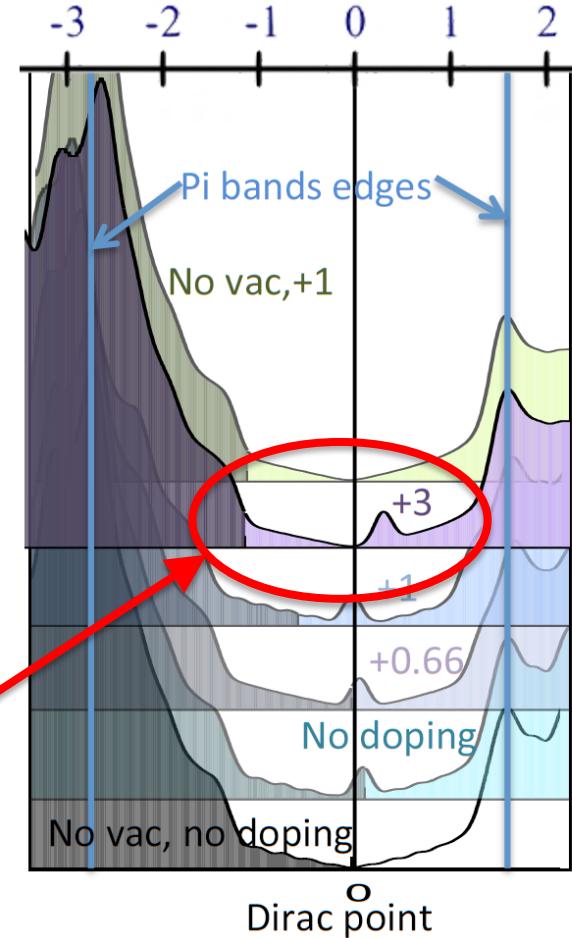
C atom on top of the vacancy



Hollow site



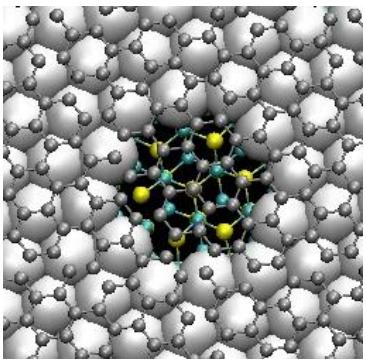
Doping of 1.1 eV shifts localized state by 0.3 eV.



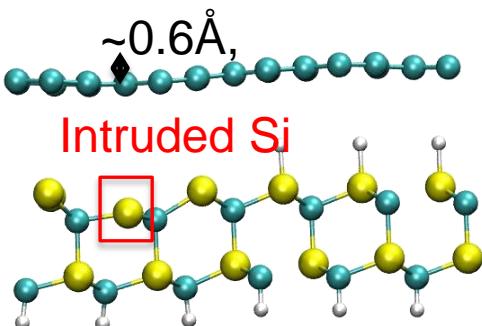
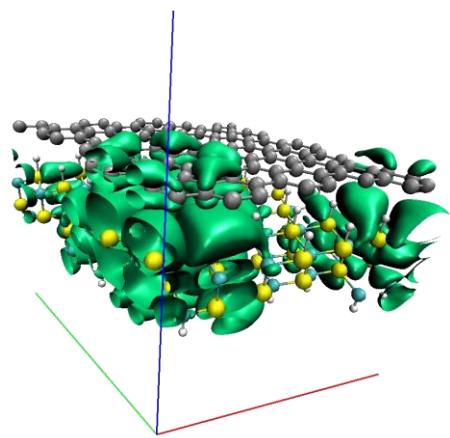
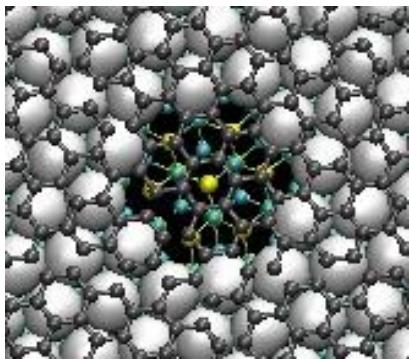
# DFT: 7-H vacancy

**7-H vacancy**

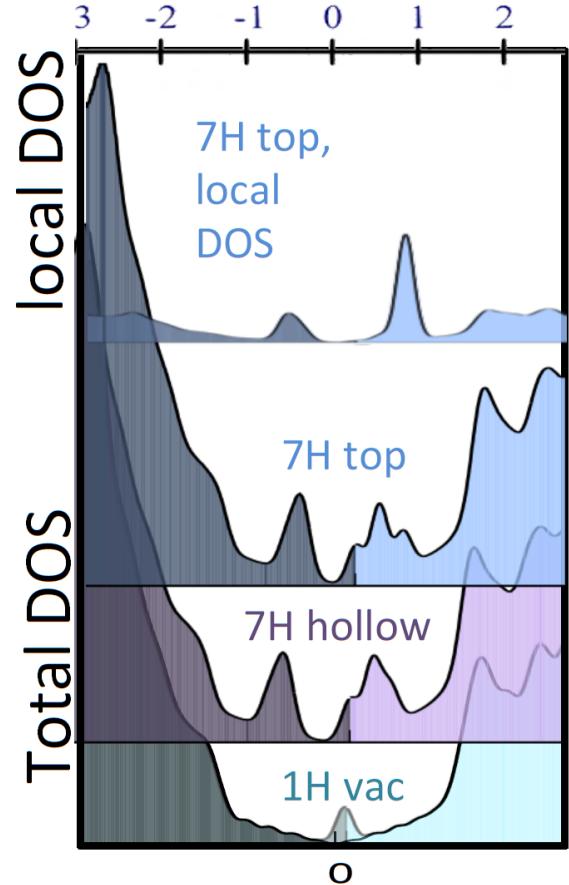
top



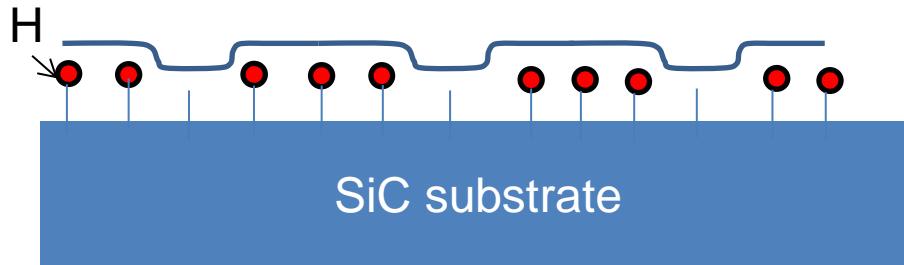
hollow



Work in progress !



# Summary



- Small dark spots ( $T_H = 600\text{-}800^\circ\text{C}$ )
- Incomplete H intercalation
- Hydrogen vacancies / Si dangling bonds
- Carrier scattering as charged impurities
- Mobility-limiting mechanism



Y. Murata



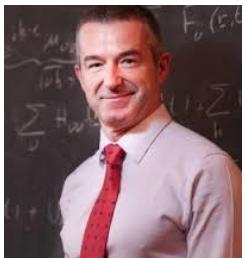
T. Mashoff



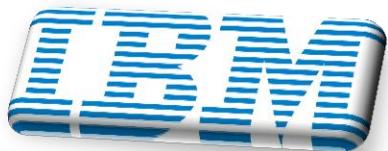
T. Cavallucci



V. Tozzini



F. Beltram



N. Pavlicek



G. Meyer



M. Takamura



H. Hibino

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# Funding

