



Nanowire week

Chamonix, France, April 25 – 29, 2022

Free-standing InSb nanoflags for quantum device applications

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Motivation

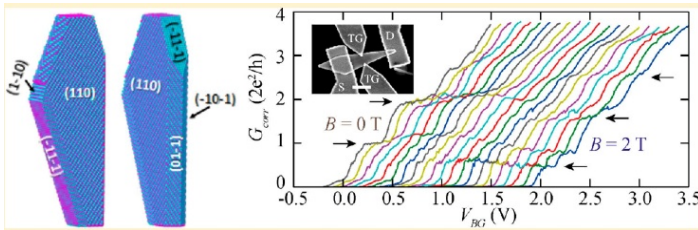
InSb:

- Small bandgap, low electron effective mass, high electron mobility, high Landè g-factor and strong spin-orbit interaction
→ promising material for optoelectronics, thermoelectrics, spintronics and quantum computing
- Large lattice mismatch with all other III-Vs → difficult integration of defect-free InSb in planar epitaxy
- Sb acts as a surfactant → difficult to control/tune the InSb morphology during the growth

Free-standing InSb (1D and 2D)

Twin-Induced InSb Nanosails: A Convenient High Mobility Quantum System

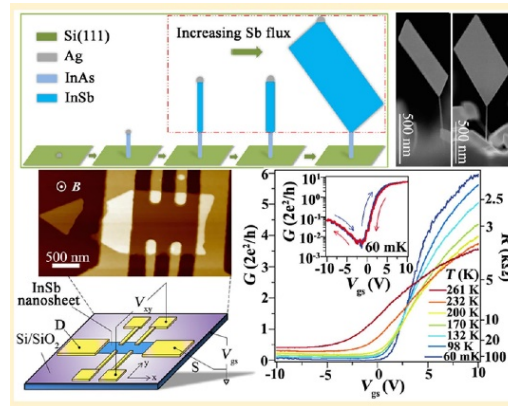
María de la Mata,[†] Renaud Leturcq,^{*,‡,§} Sébastien R. Plissard,^{||} Chloé Rolland,[‡] César Magén,[⊥] Jordi Arbiol,^{*,7,#} and Philippe Caroff^{*,§,V}



Nano Lett. 16 (2016) 825

Free-Standing Two-Dimensional Single-Crystalline InSb Nanosheets

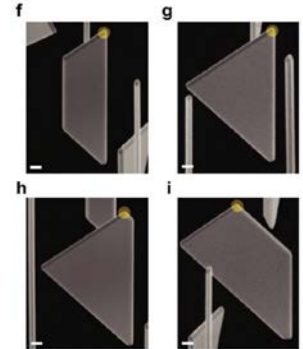
D. Pan,[†] D. X. Fan,[‡] N. Kang,[‡] J. H. Zhi,[‡] X. Z. Yu,[†] H. Q. Xu,^{*,‡} and J. H. Zhao^{*,†}



Adv. Mater. 31 (2019) 1808181

Bottom-Up Grown 2D InSb Nanostructures

Sasa Gazibegovic,^{*} Ghada Badawy,^{*} Thijs L. J. Buckers, Philipp Leubner, Jie Shen, Folkert K. de Vries, Sebastian Koelling, Leo P. Kouwenhoven, Marcel A. Verheijen, and Erik P. A. M. Bakkers



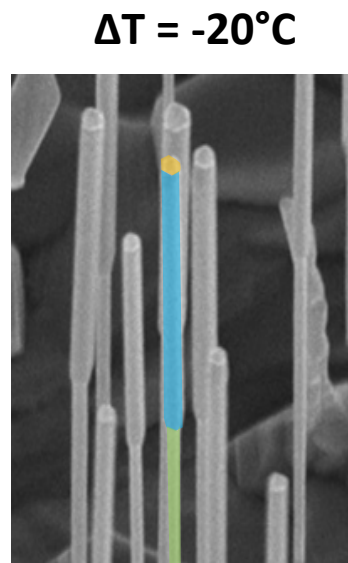
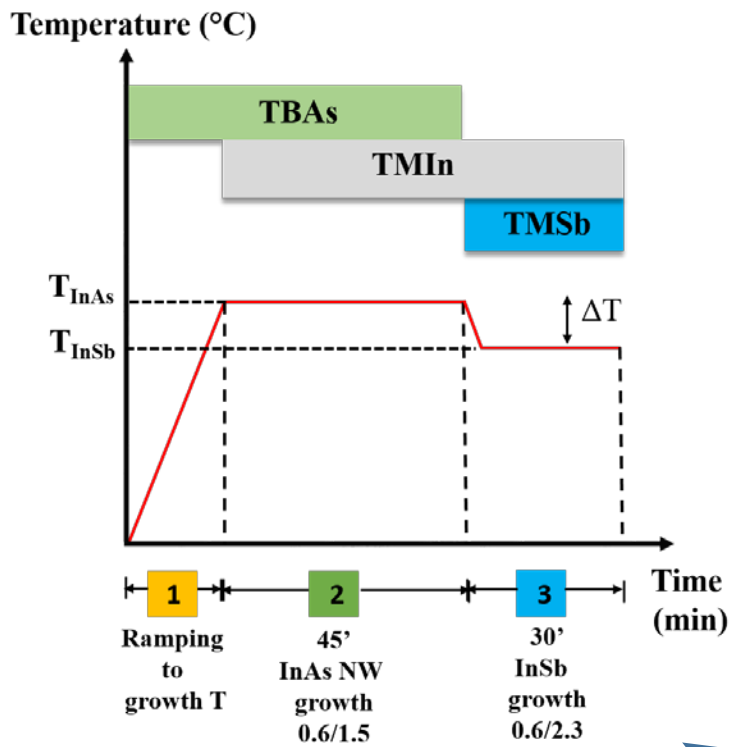
Nano Lett. 16 (2016) 834

OUR GOALS: - **morphology control** of high quality InSb nanostructures by tuning the growth parameters
- develop a directional growth protocol to achieve **free standing** InSb nanoflags (2D) on NW stems

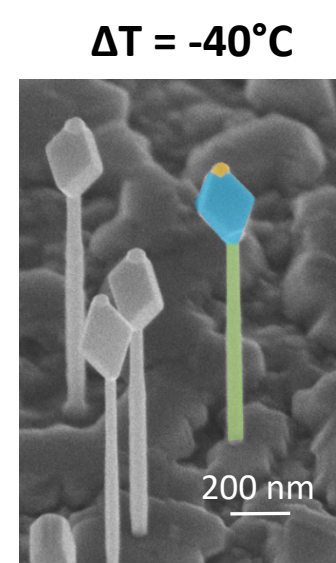
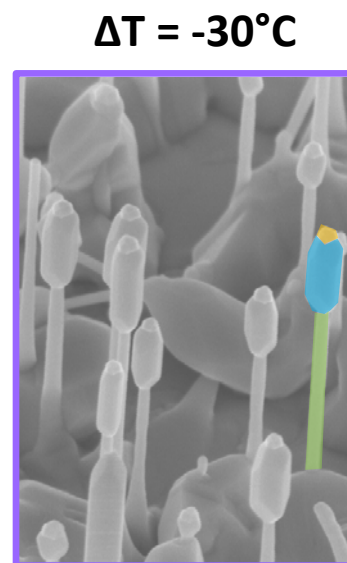
OUR APPROACH: Au assisted Chemical Beam Epitaxy using Au nanoparticle from colloidal solutions

Morphology control of InSb nanostructures

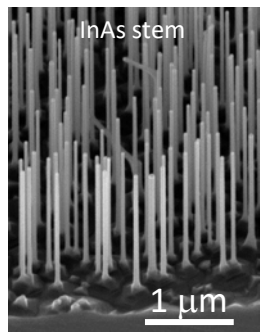
InSb growth T optimization



1D- like
Nanowires



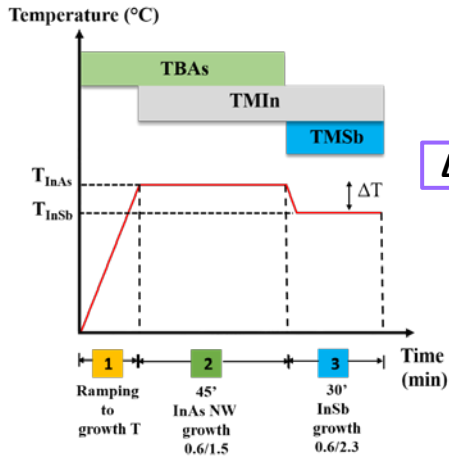
3D- like
Nanocubes



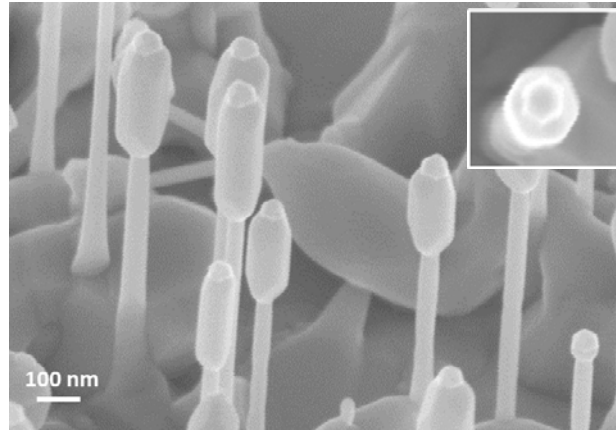
T_{InAs} : 385°C
Au NPs: 30 nm colloids
MO precursors:
TBAs, TMIIn, TMSb
Sample rotation: 5 rpm

High T_{InSb} enhances the axial growth and reduces the radial growth

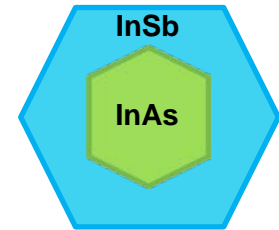
Directional InSb growth



With sample rotation uniform growth

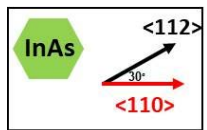


Top-view

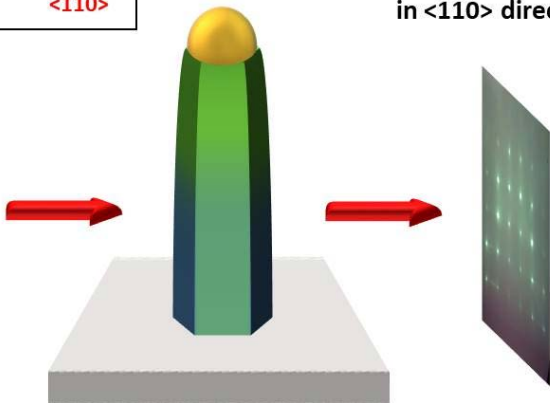


- InAs NWs have WZ crystal structure with 6 equivalent $\{112\}$ sidewalls
- InSb has ZB crystal structure with 6 equivalent $\{110\}$ sidewalls

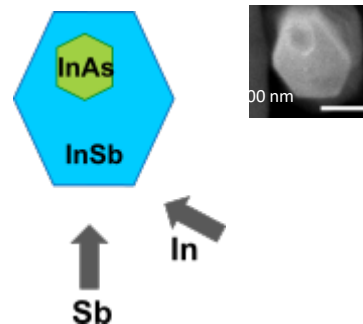
Sample alignment for directional InSb growth:



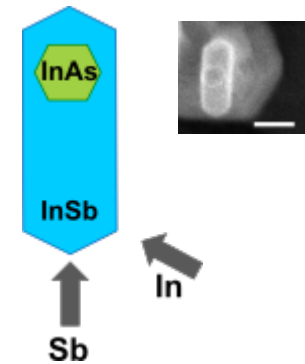
RHEED pattern of InAs NW in $\langle 110 \rangle$ direction



Sb beam projection \perp to a $\{110\}$ sidewall

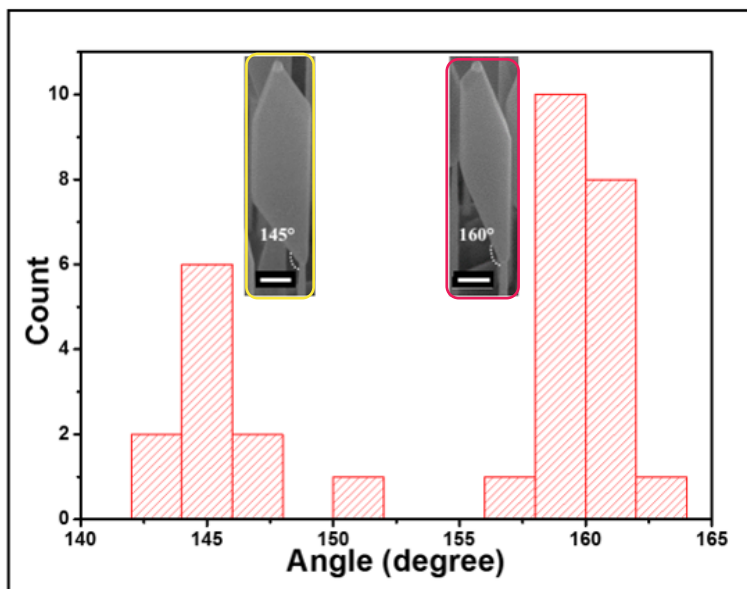
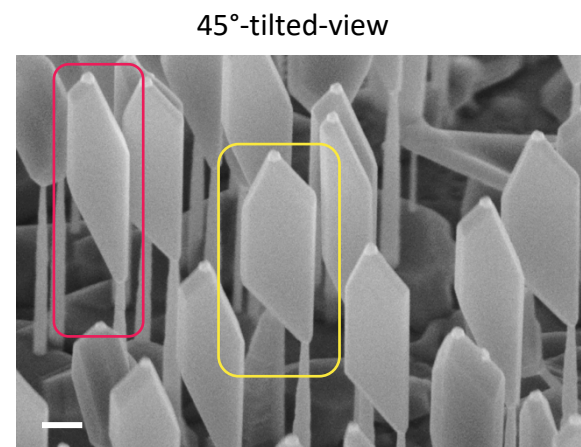
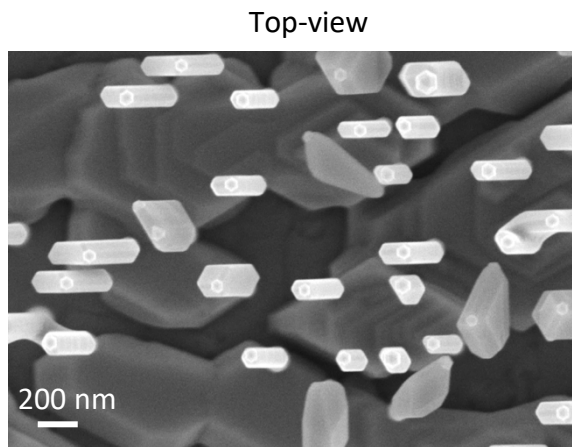
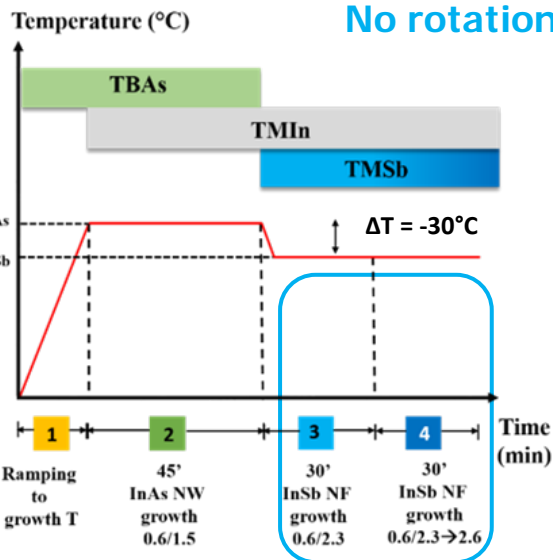


Sb beam projection \perp to a $\{112\}$ sidewall



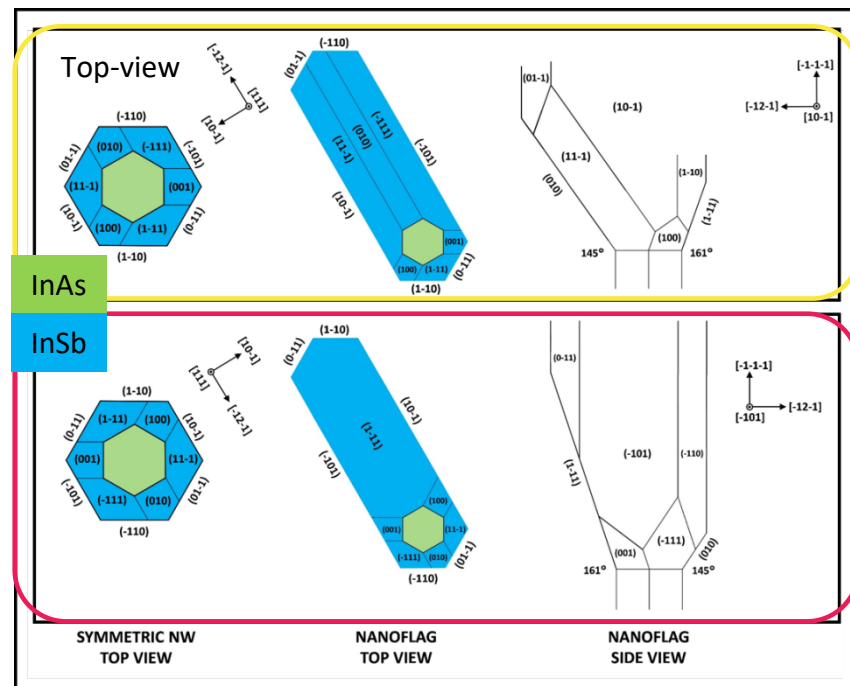
Free-standing InSb nanoflags (NFs)

No rotation, alignment, increasing Sb flux \rightarrow asymmetric growth enhancement



2 different aperture angles: 145° and 160°

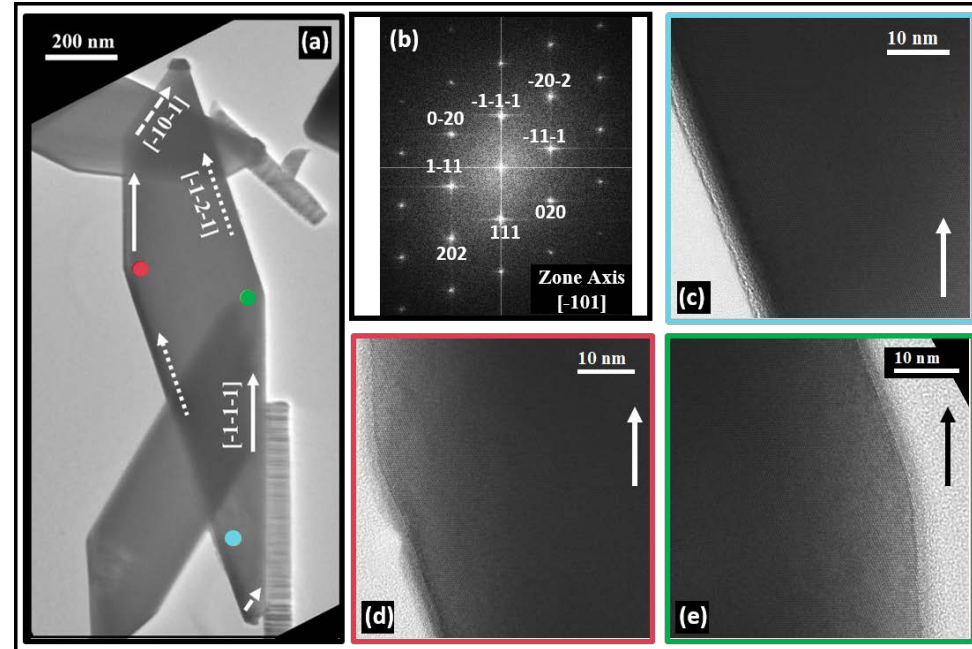
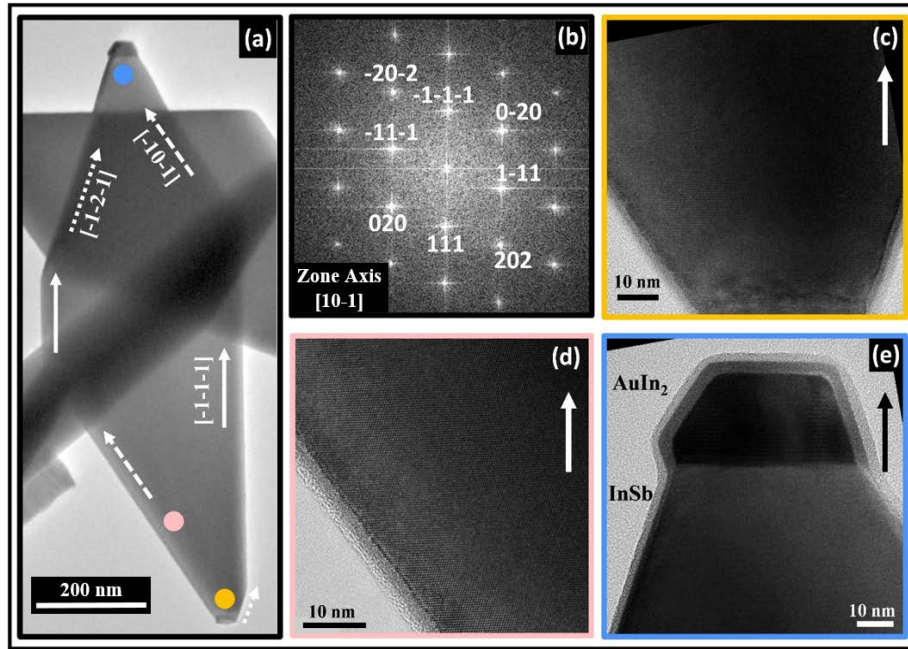
180° rotation around the growth axis of the additional facets at the interface



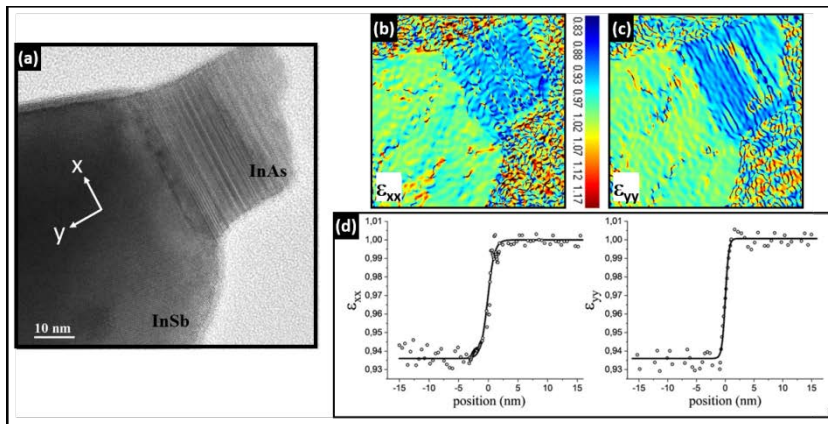
Free-standing InSb nanoflags (NFs)

145° aperture angle

160° aperture angle

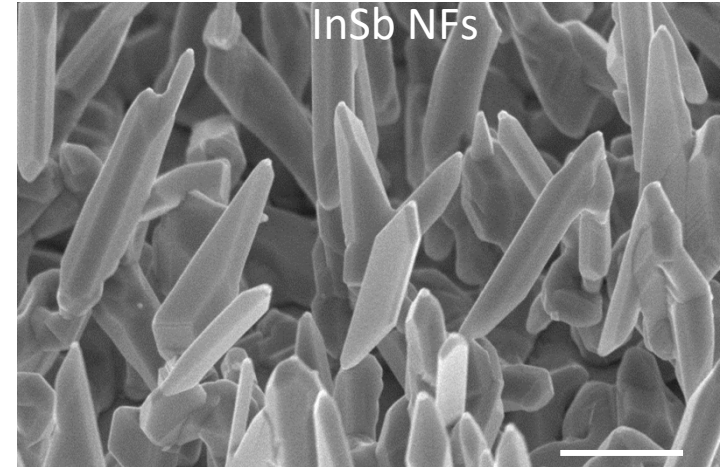
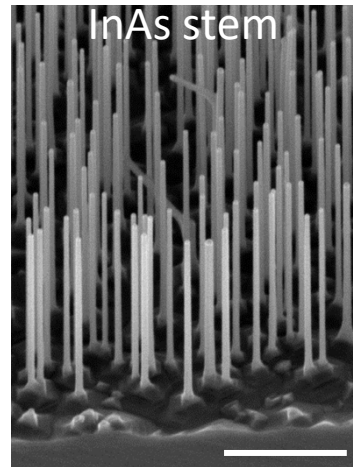
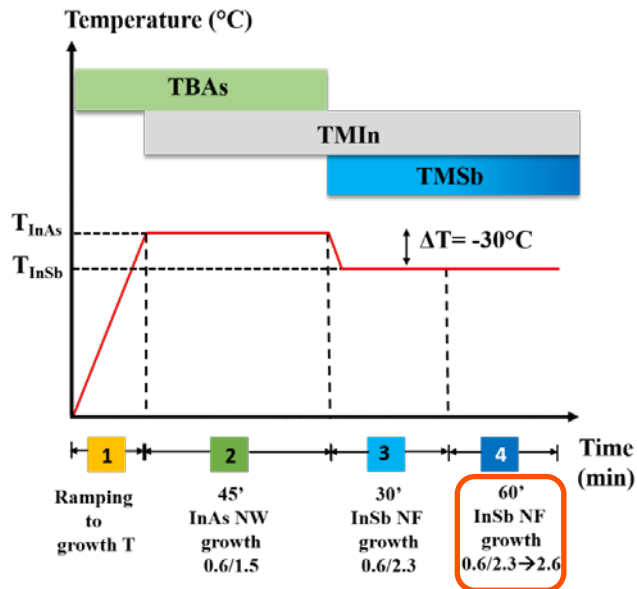


GPA

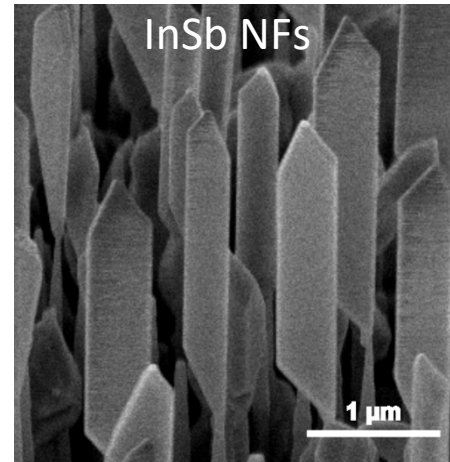
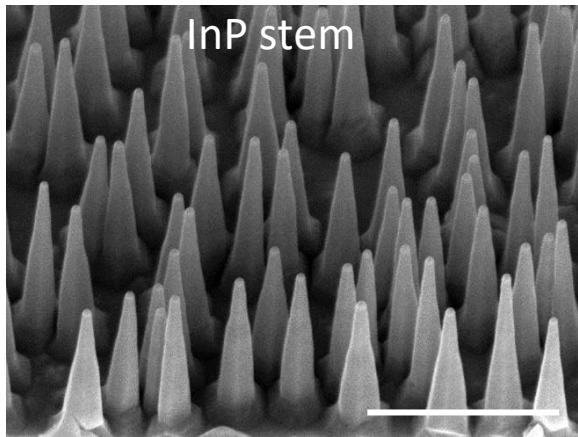


- Defect-free ZB crystal structure
- Stoichiometric composition
- Relaxed lattice parameter

Longer InSb growth time



Increasing InSb growth time

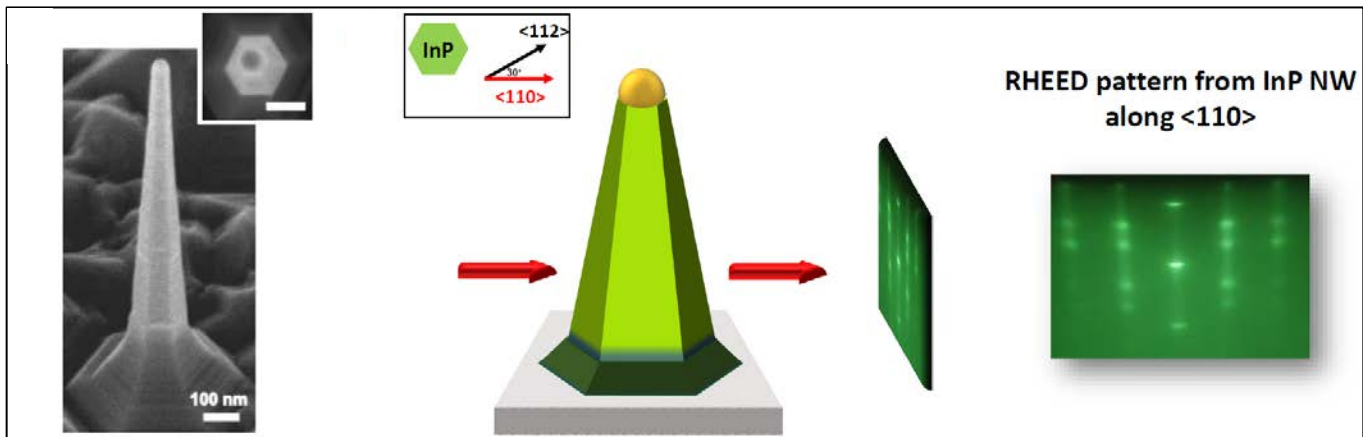


InSb NFs dimension
Length = $2.8 \pm 0.2 \mu m$
Width = $470 \pm 80 nm$
Thickness = $105 \pm 20 nm$

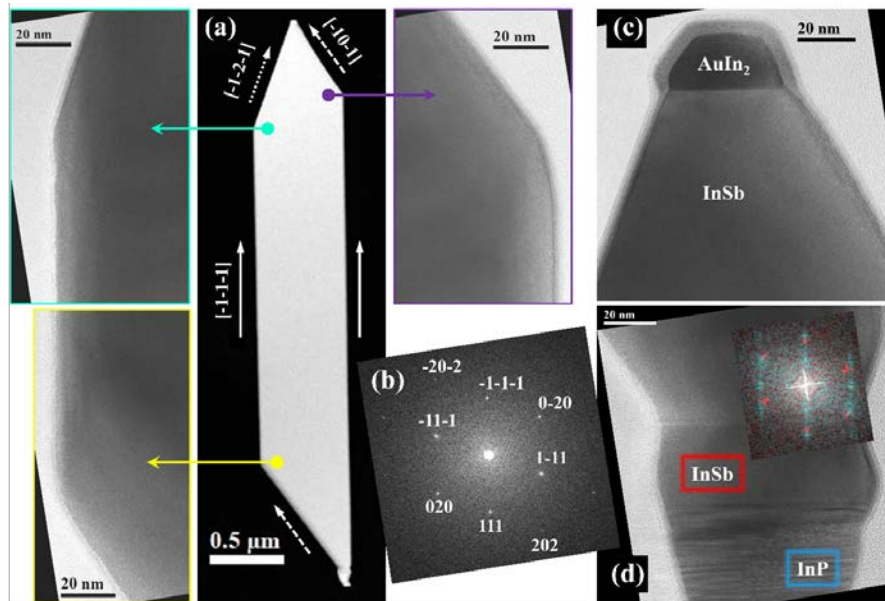
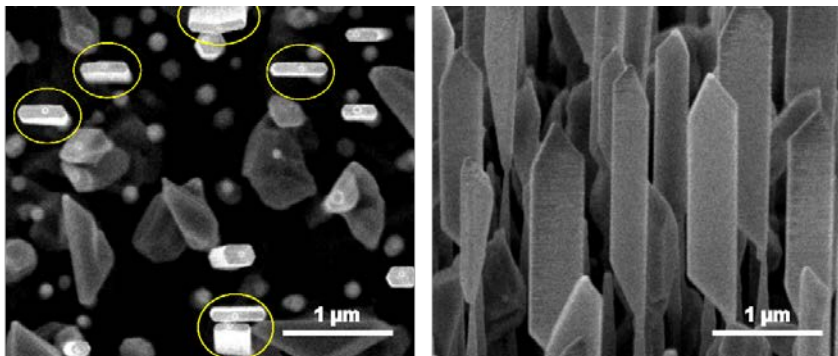
Using tapered and more robust InP NW stems, we could achieve bigger InSb NFs

Larger InSb nanoflags

InP NWs have mixed WZ/ZB crystal structure, but still $\langle 112 \rangle$ sidewalls

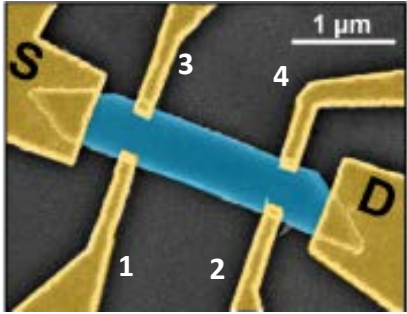


Sb beam projection \perp to a $\{112\}$ sidewall



STEM-HAADF image and corresponding HRTEM images show **defect-free InSb ZB crystal structure**.

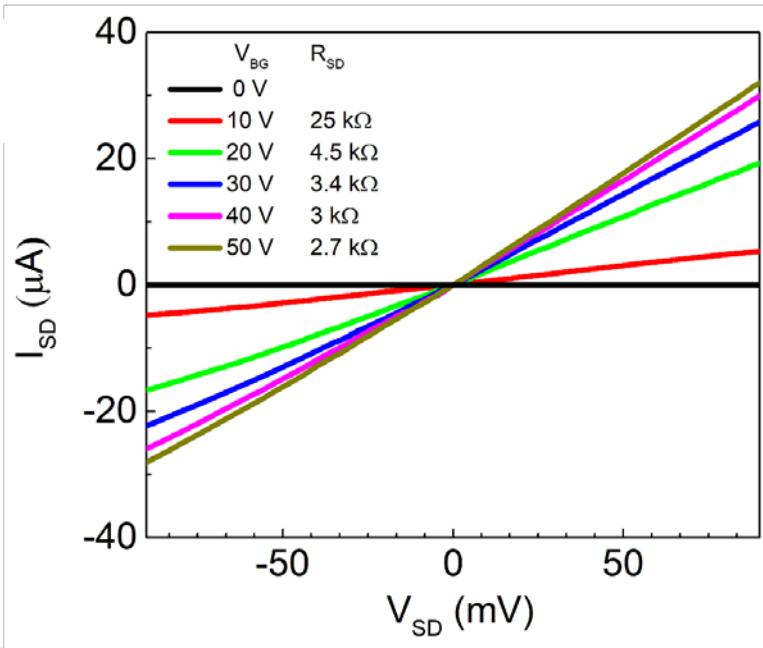
Transport measurements



Channel width (width between contacts 1-3 and 2-4)=325 nm
 Channel length (1-2 and 3-4)=1.5 μm. The NF thickness is ~100 nm.
 Contacts: 10 nm Ti/190 nm Au
 Substrate: Si/SiO₂

Hall-bar device

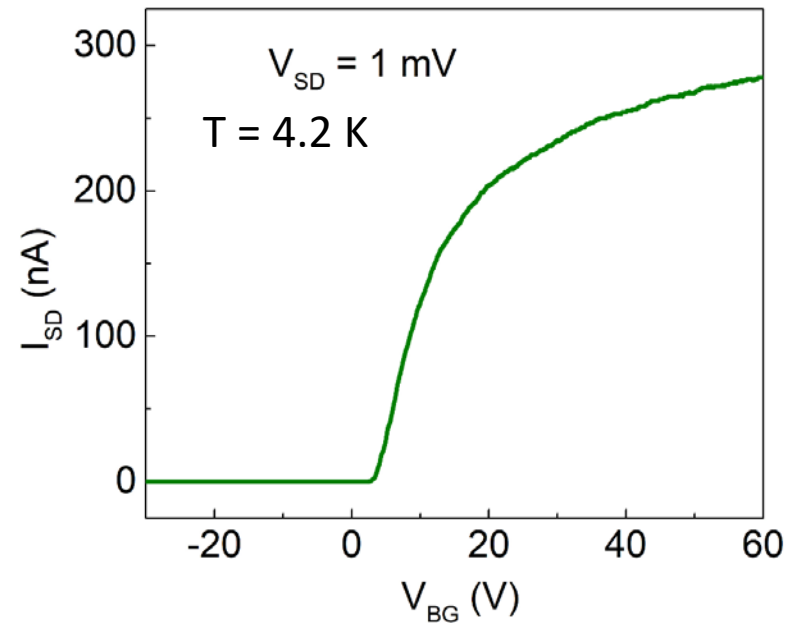
$R_{SD}(RT, V_{BG}=0 V) = 11 K\Omega$
 $R_{SD}(4.2K, V_{BG}=50 V) = 2.7 K\Omega$



Good Ohmic contacts, no Schottky barrier

n- type characteristics of InSb NFs under BG modulation

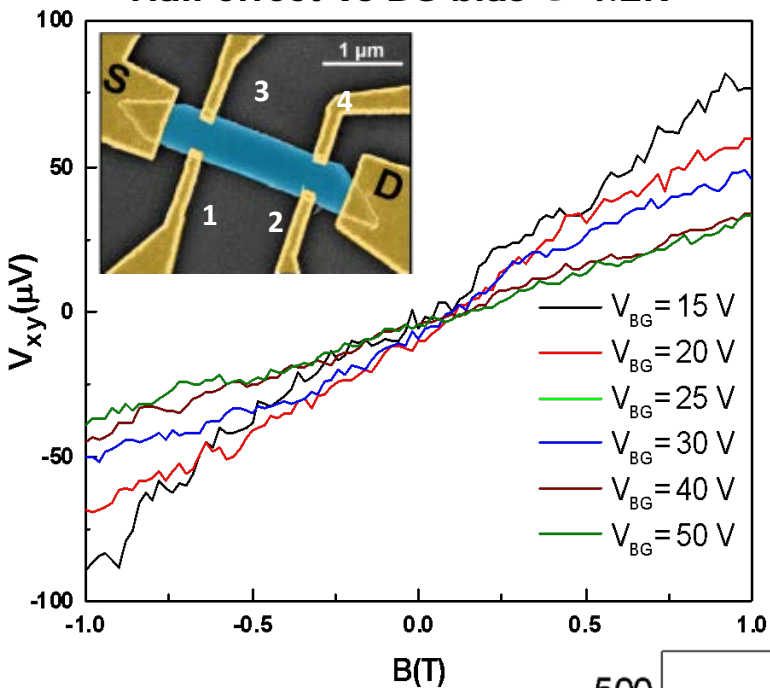
Injected current VS BG voltage @ 4.2 K



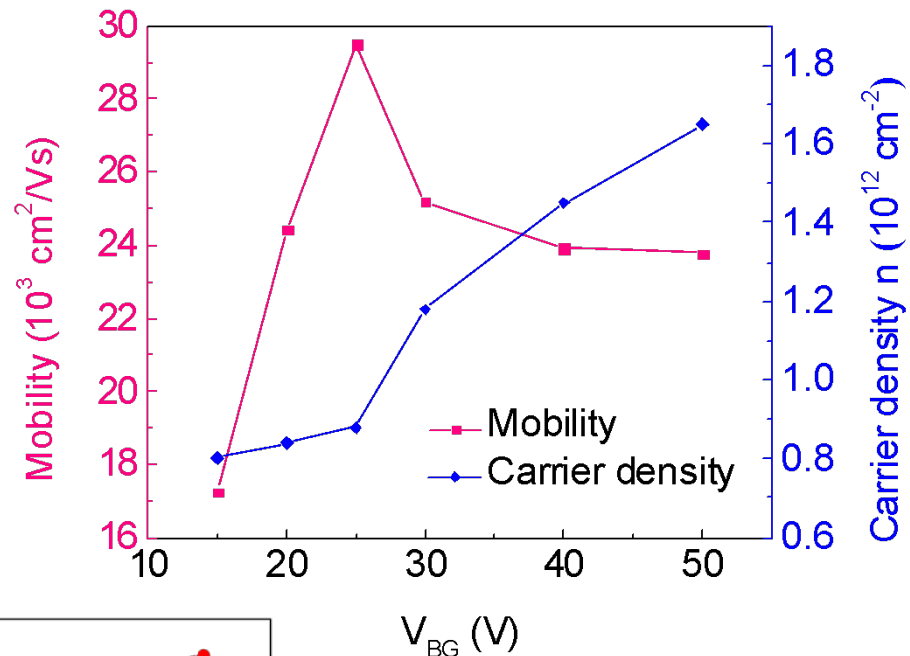
Four-probe field-effect mobility: 28000 cm²V⁻¹s⁻¹
 Nanoflags are n-type

Transport measurements

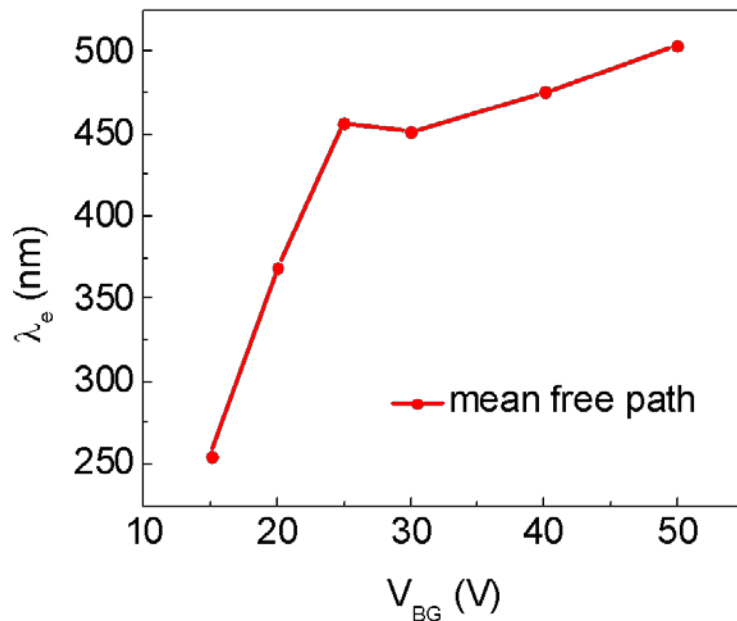
Hall effect vs BG bias @ 4.2K



Electron Hall mobility = 29500 $\text{cm}^2\text{V}^{-1}\text{s}^{-1}$

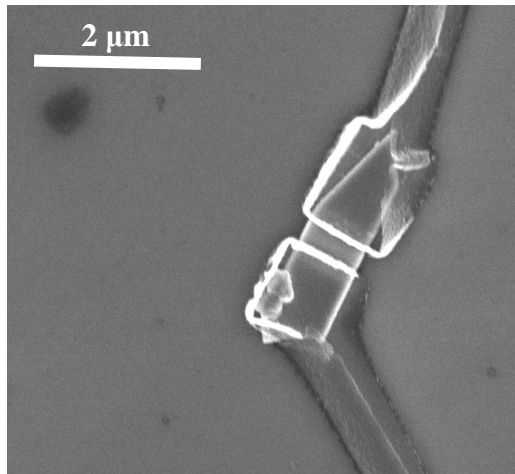


$$\lambda_e = (\hbar\mu/e)(2\pi n)^{1/2}$$



Mean free path λ_e
up to 500 nm

Ballistic InSb NFs-based Josephson junction devices

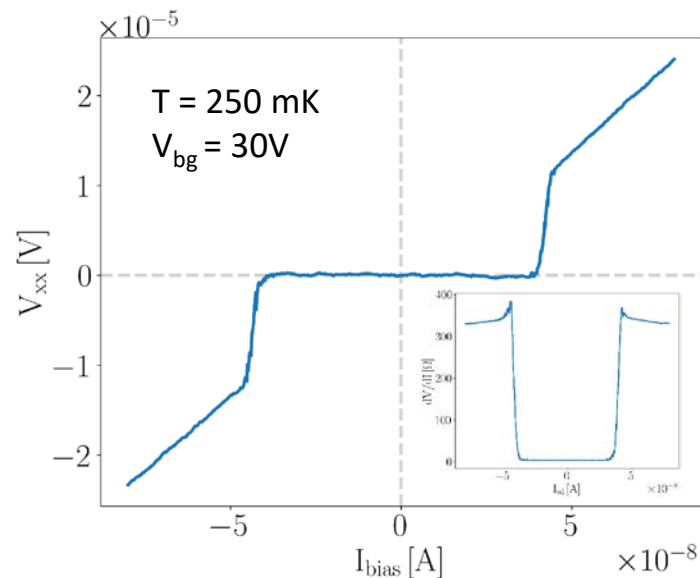


10 nm Ti/150 nm Nb
Substrate: Si/SiO₂

L = 200 nm
W = 700 nm
d = 100 nm

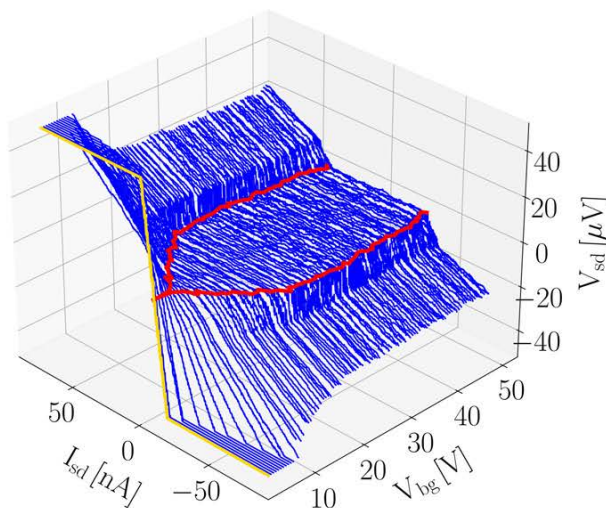
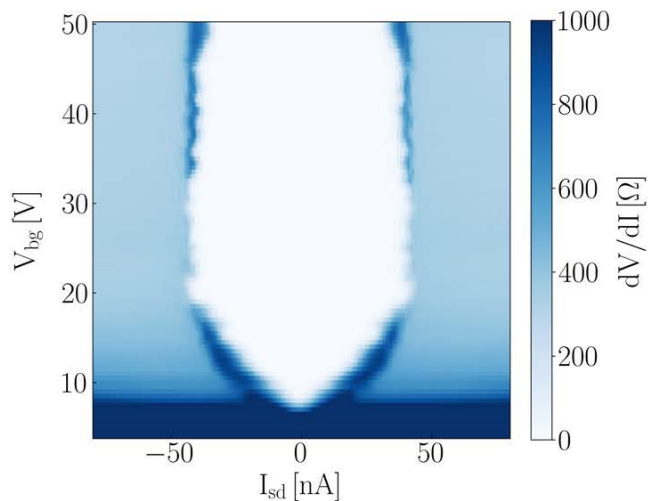
T_c = 8.44 K
Δ = 1.76 k_BT_c = 1.28 meV

Ballistic regime! Mean free path $\lambda_e >$ length L of the junction, $\lambda_e > L$.



Proximity induced superconductivity
50 nA supercurrent

Gate tunable supercurrent



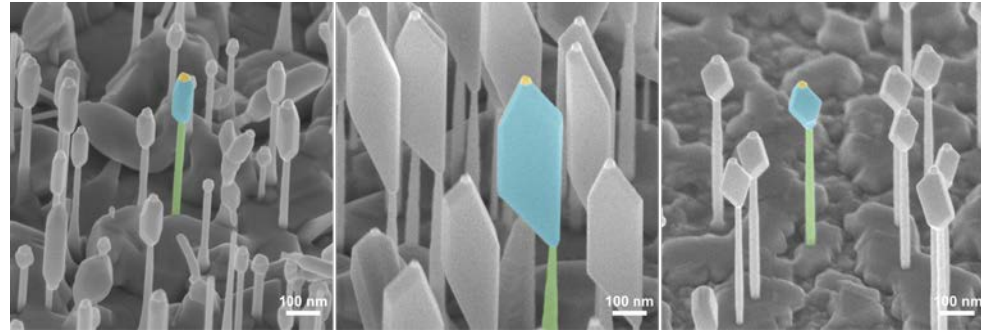
Further info:

S. Salimian et al. "Gate-controlled supercurrent in ballistic InSb nanoflag Josephson junctions"
Appl. Phys. Lett., 119, no. 21, 2021
doi: 10.1063/5.0071218

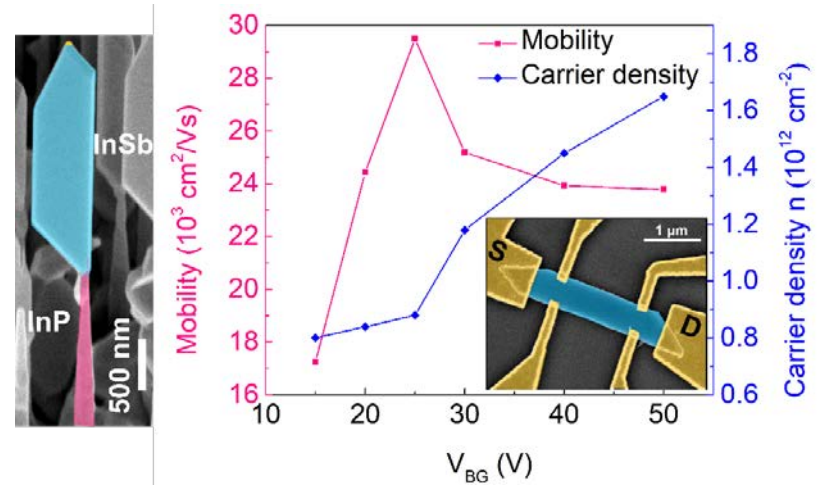
Decreasing V_{bg} below 20 V the supercurrent decreases and it disappears at 5 V

Conclusions

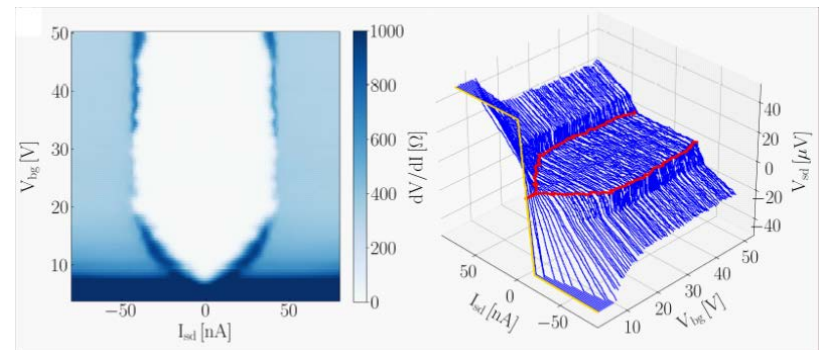
- Finding the parameters that affect axial and radial growth is important for controlling and tuning the InSb nanostructure morphology



- Using tapered NW stem and precise orientation (RHEED pattern) allow to achieve uniformly thin and large NFs
- Electron Hall mobility of about 29500 cm²/Vs
Mean free path up to 500 nm.

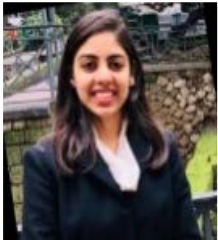


- NFs-based Josephson junction devices showed gate tunable supercurrent



- InSb NFs: versatile and convenient 2D platform for advanced quantum technologies.

Thanks to...



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Daniele Ercolani



Lucia Sorba

TEM Characterization



Francesca Rossi

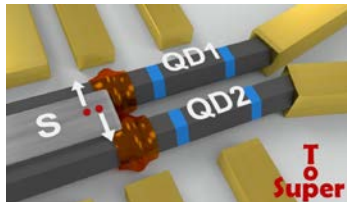
Transport Measurements



Sedighe Salimian



Stefan Heun



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Thank you for your attention!